

In-situ hierarchical micro/nanocrystals on copper substrate for enhanced boiling performance: an experimental study

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Abstract: Micro/nano structures on the Cu substrate are normally used for the enhancing the heat transfer capacity for many boiling-related applications. Conventional methods for fabricating these structures, however, require additional processing equipment and involve relatively complex processes. In this study, a simple immersion method for generating in-situ micro/nanocrystal structures on Cu substrates is developed for simplifying the fabrication with reduced cost and improving the boiling performance in the meantime. Using characterization analytical instruments including SEM, AFM, XRD, EDS and XPS, the surface morphology and chemical contents of the micro- and nanocrystals generated on the Cu substrate were examined. The experimental results showed that the hierarchical micro/nanocrystals enabled simultaneous enhancements in critical heat flux (CHF) and heat transfer coefficient (HTC), indicating the superiority of the hierarchical micro/nanocrystals in facilitating the boiling performance compared to conventional structured surfaces. It is found that the nanosheet and micro-flowers (NSMF) surface provides the largest enhancement amongst other micro/ nano structures including the nanograss forests (NG), nanograss forests and micro-petals (NGMP), and nanograss forests and micro-flowers (NGMF). Comparatively, the CHF and HTC could achieve 65.7 W/cm² and 4.9 W/cm²K, showing an increase of the CHF and HTC by 56.5 % and 170 %, respectively, using the smooth

38 surface in the same condition as the benchmark. It is implied that the liquid circulation
39 is apparently promoted through the separation of vapor-liquid pathway and the bubble
40 blanket formation is remarkably inhibited due to the special structure and morphology
41 of the hierarchical surface. However, vital factors including the surface wettability and
42 the experimental accuracy need to be considered for optimizing the boiling performance
43 in diverse scenarios.

44
45 **Key words:** Immersion method; In-situ micro/nanocrystals; Hierarchical structure;
46 liquid circulation; CHF

47 **1. Introduction**

48 Due to the significant latent heat associated with the working fluid's liquid-to-vapor
49 phase change, nucleate boiling heat transfer has been used in many high energy-
50 consuming applications, including electronic devices, nuclear reactors, lasers, and more
51 [1-3]. In this phenomenon, heat transfer coefficient (HTC), which represents the ratio
52 of heat flux to the wall superheat temperature, is used to show the heat removal capacity
53 of the device. While the critical heat flux (CHF), is regarded as the terminal safety
54 thread, which indicates a transient increase in surface temperature (approximately
55 1000 °C) at this moment, causing severe damages to devices. Advanced boiling
56 techniques that aim to concurrently improve the HTC and CHF must be developed in
57 order to prevent safety issues like burnout and dry out for increasing the limit of heat
58 transfer.

59 Previous research has demonstrated that the surface wettability and structure have a
60 considerable impact on the onset of nucleate boiling (ONB), bubble growth and
61 departure, as well as the CHF [4-8]. At low heat fluxes, hydrophobic surfaces can
62 provide an early ONB and large nucleation site density owing to the lower energy
63 barrier [9, 10]. At high heat fluxes, the formation and expansion of a vapor blanket
64 resulting from the merge of the dense bubbles prevents further enhancing the CHF.
65 However, hydrophilic surfaces can drive a small bubble diameter and a rapid bubble
66 departure frequency, as well as an effective liquid rewetting, resulting in a high CHF
67 value. Additionally, hydrophilicity is generally linked to surface morphology, which
68 can offer a large surface area and a large number of nucleation sites. The surrounding
69 liquid is effectively forced driven into the bottom of the bubbles due to the strengthened
70 capillary force on the hydrophilic surfaces, causing bubbles to quickly release before
71 merging into vapor films and increasing the effectiveness of heat removal. Nanoscale
72 structures such as nanopores [11-15], nanowires [4, 16, 17], and nanotubes [18-20] can
73 increase the capillary pressure for liquid rewetting. However, using the nanoscale
74 structures with too small pores may obstruct flow paths and decrease liquid
75 permeability, leading to the vapor bubbles coalescence. On the contrary, microscale
76 structures involving microchannels [21-23], micro-fins [24-26], micro pillars [27-30],
77 and micro porous [31-35] can facilitate liquid permeability, but reduce the capillary

78 pressure in the meantime. Thus, the combination of nano- and microscale structure
79 would be ideal to optimize the pool boiling heat transfer performance [36-40]. When
80 building a micro-/nanoscale structure, the best approach should be required to improve
81 the capillary-assisted transport of liquid and to simultaneously increase the flow paths.
82 Using this novel approach to enhance pool boiling heat transfer, the micro-/nanoscale
83 structures involving microchannel with nanowires [41-43], microchannel with porous
84 [40, 44, 45], and micro-/nanoscale porous [46, 47] have been widely investigated due
85 to their superior liquid permeability and capillary pumping capability. For example, Lee
86 et. al [42] designed a micro-nano hybrid structures that microcavities could serve as the
87 bubble nucleation sites and nanowires provided the capillary pressure for liquid
88 replenishment. This combined effects of the microcavities and nanowires can enhance
89 the CHF by delaying bubble mergence and maximizing the bubble nucleation site
90 density. Patil et. al [44] also proposed a fin-tops electrodeposited porous microchannel
91 to enhance pool boiling heat transfer through altering the bubble growth and departure
92 process. It was possible to create a micro convection so that bubbles rising from the fin
93 tops caused vigorous liquid circulation in the microchannel. Due to the controlled
94 bubble generation and departure, as well as the strengthened liquid replenishment for
95 rewetting active nucleation sites, pool boiling heat transfer process was further
96 improved. Moreover, hierarchical structures that have the artificial microcavities and
97 micro-/nano pores can significantly optimize the boiling heat transfer performance
98 because of the improved bubble nucleation, strengthened capillary pressure as well as
99 increasing flow paths on the nanoengineered surfaces. Li et. al [48] proposed an ultra-
100 scalable three-tire hierarchical Cu nano engineered surfaces by using electrochemical
101 deposition for optimizing the boiling heat transfer. Through multiple-pronged strategy,
102 including raising the nucleation site density, regulating bubble behavior and expanding
103 liquid-vapor separation pathways, the structure length scale can be adjusted to enhance
104 pool boiling heat transfer. For improving boiling heat transfer, Wen et. al [49] also
105 created a two-level hierarchical Cu nanowires framework with patterned nanowire
106 arrays. Long nanowire arrays' capillary pumping can act as continuous feeder channels
107 to rewet active nucleation sites, increasing the CHF, while microcavities between short
108 nanowires can provide high-density nucleation sites for increasing the HTC and
109 decreasing the ONB.

110 However, most fabrication methods such as electrochemical deposition, chemical
111 vapor deposition, chemical etching and sintering of the above-mentioned micro/nano
112 structures are complex, which restrain their effective applications in different
113 conditions. Due to its ease of use, low cost, and environmental friendliness, the solution
114 immersion technique has recently been regarded as a viable method for producing
115 nanostructures on metal surfaces. These surfaces also have potential in applications for
116 pool boiling because of chemical stability and endurance under diverse serve
117 circumstances. For instance, Kim et. al [50] fabricated super-hydrophilic aluminum
118 surfaces through the solution immersion method, and explored the effects of surface
119 roughness on the pool boiling heat transfer performance. As reported, the CHF was
120 positively connected with the roughness within a particular range, but when the

121 roughness beyond that range, the hydrodynamic limit took effect. Boehmite
122 nanostructures were discovered to have considerable advantages in improving the CHF
123 and surface wettability [51]. It was found that, the aluminum surface induced an
124 enhanced CHF ascribed to the highly stable super-hydrophilic properties. Moreover,
125 CuO nanostructures using solution immersion method also has been tried to enhance
126 pool boiling heat transfer. For example, Im et. al [52] presented low-profile, flower-like
127 CuO nanostructures on a smooth surface and a microgroove surface, respectively. The
128 results showed that the CHF was improved by at least 58% in different scenarios due
129 to the increasing capillary wicking induced by their high surface area-to-volume ratio.
130 Xie et. al [53] investigated the effect of the surface wettability on the boiling
131 enhancement. The maximum CHF was increased by 68% by fabricating a dual-layer
132 structure using nanograss as the substrate and micro flowers with varying covering
133 densities as the superstructures. In addition, they have also looked into the combined
134 effects of surface wettability and patterned nanostructures on boiling heat transfer [54].
135 The results showed that the boiling curves are shifted to the left by patterned
136 nanostructured surfaces, and the CHF rises as the density of the nanograss cover
137 increases. Li et. al [55] investigated the wicking degeneration on nanoengineered
138 surfaces and clarified the significance of absorbing ambient volatile organic compounds
139 from air. The hierarchical micro/nanostructures that were highly scalable, conformal,
140 cost-effective and with exceptional wicking abilities displayed great pool boiling heat
141 transfer. Although previous studies have contributed to a basic understanding of the
142 mechanism for improving boiling heat transfer based on micro- and nanocrystal
143 structures, more analysis of the impact of crucial factors like the immersion time on
144 surface modification and thermal promotion of nanoengineered surfaces is worthy to
145 be investigated.

146 In this work, Cu surfaces were treated with solution immersion, a process that is both
147 economic and environmentally benign, to produce nanocrystals. The formation process
148 of in-situ growth of micro/nanocrystals on Cu surfaces is investigated. The
149 morphological and chemical characteristics were characterized, and the impact of
150 immersion time on surface wettability and morphology were investigated. Based on
151 these, boiling heat transfer experiments has been done on nanoengineered surfaces with
152 the in-situ hierarchical Cu micro/nanocrystals, by which significant thermal
153 improvement has been achieved in wide-range experimental conditions.

154 **2. Experimental sections**

155 **2.1 Sample fabrication and characterization methods**

156 The manufacturing of boiling surfaces in this work involves three basic steps:
157 washing, chemical alteration of surfaces, and vacuum drying. The whole copper
158 surfaces of $20\text{mm}\times 20\text{mm}\times 3\text{mm}$ were polished with 2000 grid SiC sandpaper for
159 removing the impurities on the surfaces. The polished copper surfaces were then
160 scrubbed with an ultrasonic acetone solution for 15 minutes to get rid of the residue.
161 Then, the polished surfaces were rinsed with the ultrasonic ethanol and DI water for 10

162 min, respectively. Finally, these copper surfaces were placed in an oven and heated at
163 65 °C for 10 min.

164 The experiment materials, sodium hydroxide (NaOH, AR, $\geq 96.0\%$) and ammonium
165 persulfate ((NH₄)₂S₂O₈, AR, 98.0%) were purchased from the Sinopharm Chemical
166 Reagent Co., Ltd. A 250 ml glass beaker holding the 100 ml aqueous solution of 2.5 M
167 NaOH and 0.1 M (NH₄)₂S₂O₈ was used to prepare the 100 mL aqueous solution. The
168 dried and polished copper surfaces were placed into the above solution at the room
169 temperature for 3 min, 10 min, 60 min and 120 min, respectively. The reaction
170 conditions are summarized in **Table 1**. The initial colorless solution in the glass beaker
171 became increasingly blue. The manufactured copper surfaces were rinsed with DI water
172 after the chemical alteration in the beaker and put into the vacuum oven to be heated at
173 60 °C to remove the leftover solution. After the chemical modification and vacuum
174 drying, blue or dark blue colour appeared on the copper surfaces.

175 **Table 1.** Reaction conditions for the nanocrystals on the Cu substrate.

Sample	Concentration of (NH ₄) ₂ S ₂ O ₈ (M)	Concentration of NaOH (M)	Reaction time (min)
1			3
2			10
3	2.5	0.1	60
4			120

176 Scanning electron microscopy (SEM, Gemini 300, ZEISS, Germany) and atomic
177 force microscopy (AFM, Dimension Edge, Bruker) techniques were applied to clearly
178 identify the surface morphology properties of the Cu micro/nanocrystals and to
179 investigate the growing thickness of the reaction products. Meanwhile, the energy-
180 dispersive X-ray spectroscopy (EDS, INCA X-Act, Oxford) was used to get a
181 quantitative analysis (Cu, O, C) of the reaction products on the Cu substrate. An X-ray
182 photoelectron spectrometer (XPS, ESCALAB 250xi, America) was used to examine
183 the sample's XPS spectra.

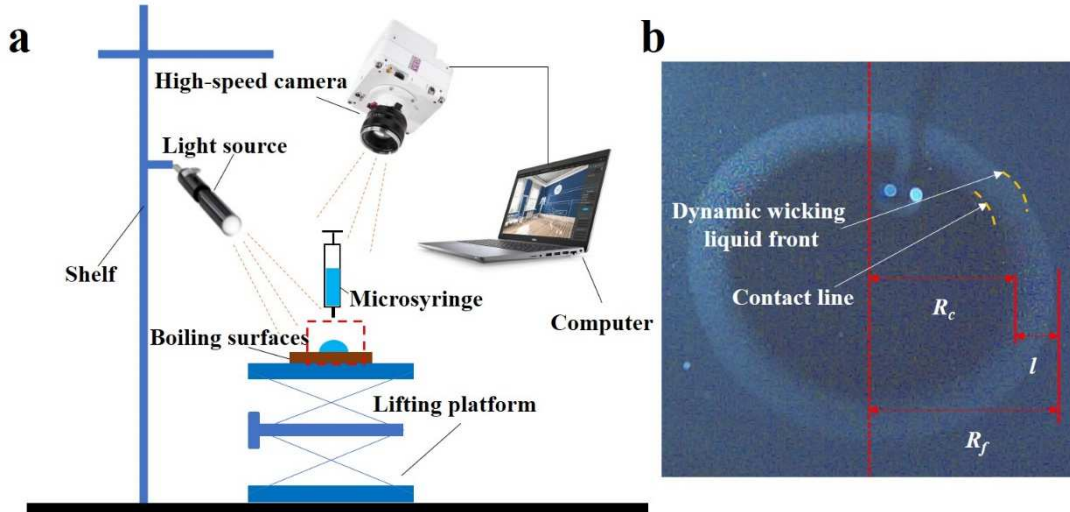
184 2.2 Wicking capability test system

185 **Figure 1a** shows the wicking capability test system including a micro-syringe pump,
186 a lifting platform, a high-speed camera connecting with the computer and a light source.
187 In this experiment, 1 μ L DI-water droplet obtained from the micro syringe pump were
188 released on the surfaces of samples. To observe the wicking phenomenon, a high-speed
189 camera (Phantom, VEO 410) was set to record 1 s at 1000 fps for capturing the droplet
190 spreading process. The micro-syringe was positioned 3 mm away from the boiling
191 surfaces in order to minimize mistakes when the droplets drop onto the boiling surfaces.
192 **Figure 1b** shows the captured image of droplet spreading and wicking. R_f represents
193 the capillary wicking radius (from center to dynamic wicking liquid front) and R_c
194 represents the droplet spreading radius (from center to contact line). The wicking
195 distance l is the distance between R_f and R_c . The wicking distance can be defined as

196 follows [56]:

197
$$l = \sqrt{\frac{2P_{cap}}{K}} t = W\sqrt{t} \quad (1)$$

198 where P_{cap} is the capillary pressure, K is the permeability and W is the wicking
199 coefficient. Through analyzing the high-speed images, the wicking distance can be
200 measured and wicking coefficient can be calculated according to equation (1).



201

202 **Figure 1.** (a) Schematic of wicking capability experiment, (b) Captured high-speed
203 camera images

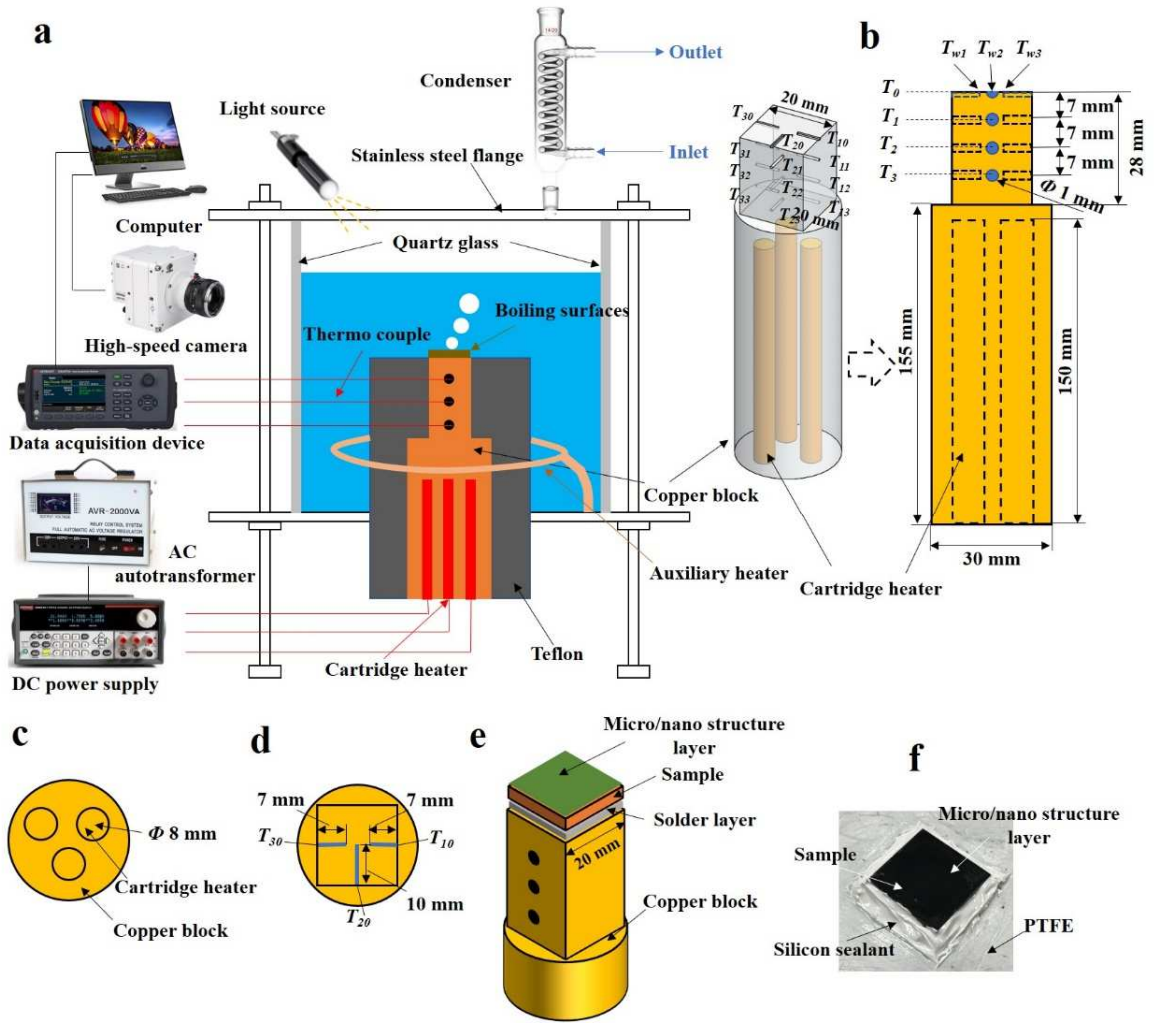
204 2.3 Pool boiling experimental setup

205 **Figure 2a** shows the schematic diagram of pool boiling experimental setup. It
206 consists of a boiling chamber, a heating system, an auxiliary heating system, a
207 condensing system, a viewing system, a data gathering system. The working fluid was
208 DI water, and the boiling chamber was constructed of quartz glass with a thickness of
209 5 mm. To keep the saturation of the DI water under 1 atm, an auxiliary heater that is
210 submerged in the DI water was attached around the PTFE block. A spiral tube was
211 added on the boiling chamber for condensing the water vapor in order to maintain the
212 liquid level of the DI water. The main body of the heating system was a copper block
213 with three cartridge heaters embedded in the cylindrical bottom. And the copper block
214 was integrated into the PTFE to maintain one-dimensional heat transfer. The cartridge
215 heaters were connected to the DC power supply, and the DC power supply controlled
216 by the AC autotransformer was responsible for the heating power of cartridge heaters.

217 As shown in **Figure 2a** and **b**, the copper block had a height of 183 mm height and
218 the upper of copper block is a square column with a width of 20 mm. To measure the
219 heat flux and temperature gradients in the vertical direction, nine holes with a diameter
220 of 1 mm were drilled along the copper column at 7 mm intervals for placing the T-type
221 thermocouples. These nine thermocouples were distributed on three planes of the

222 copper block, with three thermocouples on each plane. As shown in **Figure 2b** and **d**,
223 the thermocouples (T_{21} , T_{22} and T_{22}) were placed in the holes with a depth of 10 mm
224 along the center line of the copper block, which can be used to calculate the temperature
225 gradient of the copper block. Also, the thermocouples (T_{11} , T_{12} , T_{13} , T_{31} , T_{32} and T_{33})
226 were placed in the holes with a depth of 7 mm, which can be used to calculate the
227 surface temperature. **Figure 2b** shows the thermocouples' distribution. Another three
228 thermocouples (T_{01} , T_{02} and T_{03}) were embedded on the top of the copper column
229 (**Figure 2d**). The average temperature of each plane was represented by T_0 , T_1 , T_2 and
230 T_3 from top to the bottom, respectively. Meanwhile, the temperature of DI water was
231 measured by the thermocouple T_m . The data acquisition system (Keysight, DAQ 970A)
232 recorded all the T-type thermocouples' temperature readings. Moreover, the high-speed
233 camera also recorded bubble dynamics including nucleation, growth and departure.

234 As shown in **Figure 2e**, the sample was welded on the copper block using lead-free
235 solder. First, the solder was melted by the soldering iron and cooled down to form a
236 solder layer on the copper substrate. Second, controlling the power of DC power supply
237 to heat the copper block until the surface temperature T_0 reaches the solder layer melting
238 point of 227 °C. Third, the sample was pressed on the solder layer (0.1 mm thick) and
239 a heavy object was placed on the sample until the surface temperature returns to the
240 room temperature. The contact thermal resistance of solder layer can be neglected
241 because it only accounts for less than 1% of the total resistance [57], which also has
242 been proved in the existing research [35, 58]. In addition, to prevent the heat losses
243 from the gap between the PTFE and sample, a silicon sealant was used to fill this gap
244 on the surface border (**Figure 2f**).



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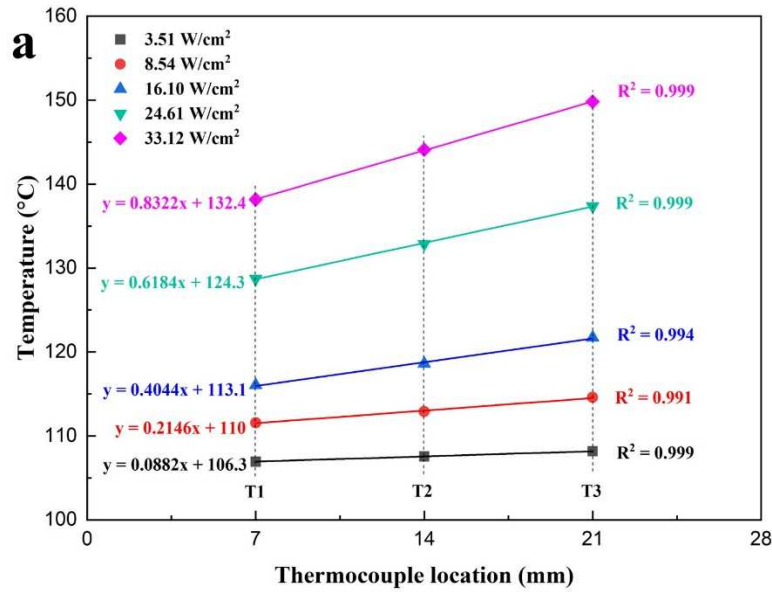
246 **Figure 2.** (a) Schematic diagram of pool boiling experimental setup (b) Front view
 247 of copper block (c) Distributions of the cartridge heater inside copper block (d) Top
 248 view of copper block (e) Sample installation (f) Sample with silicon sealant

249 **2.4 Test procedures and data reduction**

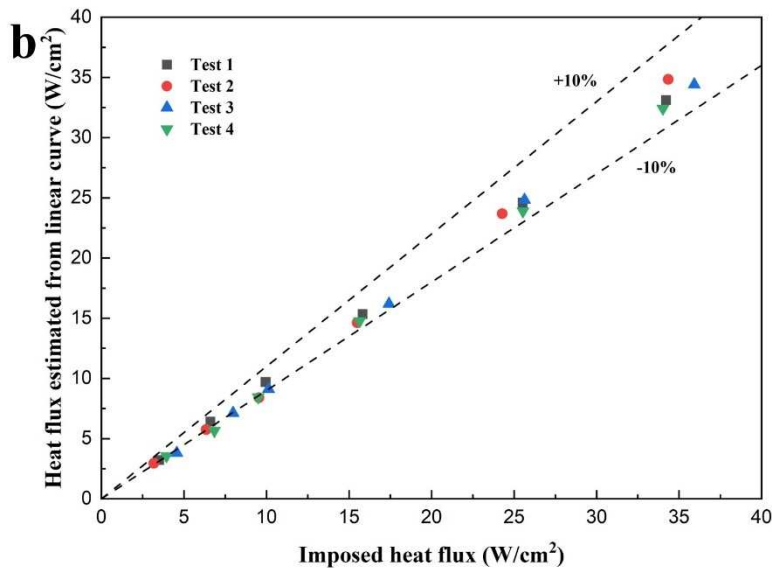
250 Prior to the pool boiling tests, DI water was heated by the auxiliary heater to the
 251 saturation temperature ($T_{sat} = 100\text{ }^{\circ}\text{C}$) for roughly 60 min to release extra gas. The
 252 voltage was then gradually increased to the cartridge heaters in order to improve the
 253 heat flux of the boiling surfaces. The temperature reached a stable level following each
 254 incremental voltage increase after around 15 min. Every 1 minute, the temperature was
 255 measured, and each thermocouple's temperature variation was less than $\pm 0.2\text{ }^{\circ}\text{C}$. When
 256 the surface temperature exhibited the significant rise at one point, the CHF phenomenon
 257 could occur. W In order to protect the PTFE block, the experimental devices were
 258 turned off after the CHF reached.

259 Before calculating the heat flux, one-dimensional heat transfer along the copper
 260 block should be verified. **Figure 3a** shows the temperature distribution (T_1 , T_2 and T_3)

261 along the copper block measured by the thermocouples at different heat fluxes. As
 262 shown in **Figure 3a**, the R-square values obtained from the linear curves fitting are all
 263 higher than 0.99. Thus, the radial heat loss from the copper block can be neglected and
 264 the heat flux calculated from the T_1 , T_2 and T_3 is effective. Otherwise, the linear curve
 265 cannot fit the experimental data. **Figure 3b** show the comparison between the imposed
 266 heat flux based on the corresponding heating power of cartridge heaters and the heat
 267 flux estimated from the linear curve according to the experimental data from 4 tests of
 268 smooth surface, indicating that the heat losses are less than 10%.



269



270

271 **Figure 3.** (a) Temperature distribution at different heat fluxes (b) Comparison
 272 between the imposed heat flux corresponding to heating power of cartridge heaters
 273 and heat flux estimated from linear curve

274 By one-dimensional Fourier's law, the input heat flux q can be calculated as follows:

275 $q = -\lambda_{Cu} \frac{dT}{dz}$ (2)

276 where λ_{Cu} is the thermal conductivity of the pure Cu, dT/dz is the axial temperature
 277 gradient of the Cu block determined by the three thermocouple readings, Δz is the
 278 distance between two thermocouple locations.

279 Utilizing the three points backward Taylor's series approximation, the temperature
 280 gradient in the copper block was determined.

281 $\frac{dT}{dx} = \frac{3T_{21} - 4T_{22} + T_{23}}{2\Delta x}$ (3)

282 where T_{21} , T_{22} and T_{23} are the temperatures along the center line of the copper block.

283 The local surface temperatures can be calculated based on the Fourier's theory and
 284 uniform intervals as follows:

285 $T_{w1} = T_{11} - \frac{T_{13} - T_{11}}{2}$ (4)

286 $T_{w3} = T_{31} - \frac{T_{33} - T_{31}}{2}$ (5)

287 Also, T_{w2} can be calculated based on the temperature gradient through T_{21} , T_{22} and T_{23} .
 288 Therefore, the surface temperature T_w can be calculated by average value $(T_{w1} + T_{w2} +$
 289 $T_{w3})/3$.

290 Herein, the HTC (h) can be determined as follows:

291 $h = \frac{q}{T_w - T_{sat}}$ (6)

292 where Δz_0 is the distance between the top surface, T_w is the boiling surface bottom
 293 temperature; T_{sat} is the saturated temperature of the working fluid.

294 2.5 Uncertainty analysis

295 The absolute errors for the calibrated T-type thermocouples were 0.2 °C. Copper's
 296 thermal conductivity was calculated to be 398 W/m·K, and the thermocouple location
 297 errors were determined to be 0.2 mm. The uncertainties of heat flux, heat transfer
 298 coefficient, and superheat were calculated using conventional method as follows in
 299 accordance with the error-propagation law:

300 $\frac{U_q}{q} = \sqrt{\left(\frac{U_{\lambda_{Cu}}}{\lambda_{Cu}}\right)^2 + \left(\frac{3U_{T_1} \cdot \lambda_{Cu}}{\Delta z \cdot q}\right)^2 + \left(\frac{4U_{T_2} \cdot \lambda_{Cu}}{\Delta z \cdot q}\right)^2 + \left(\frac{U_{T_3} \cdot \lambda_{Cu}}{\Delta z \cdot q}\right)^2 + \left(\frac{U_{\Delta z}}{\Delta z}\right)^2}$ (7)

$$301 \quad \frac{U_{\Delta T}}{\Delta T} = \sqrt{\left(\frac{U_{T_1}}{T_1}\right)^2 + \left(\frac{U_q}{q}\right)^2 + \left(\frac{U_{\lambda_{Cu}}}{\lambda_{Cu}}\right)^2 + \left(\frac{U_{\Delta z}}{\Delta z}\right)^2 + \left(\frac{U_{T_{sat}}}{T_{sat}}\right)^2} \quad (8)$$

$$302 \quad \frac{U_h}{h} = \sqrt{\left(\frac{U_q}{q}\right)^2 + \left(\frac{U_{\Delta T}}{\Delta T}\right)^2} \quad (9)$$

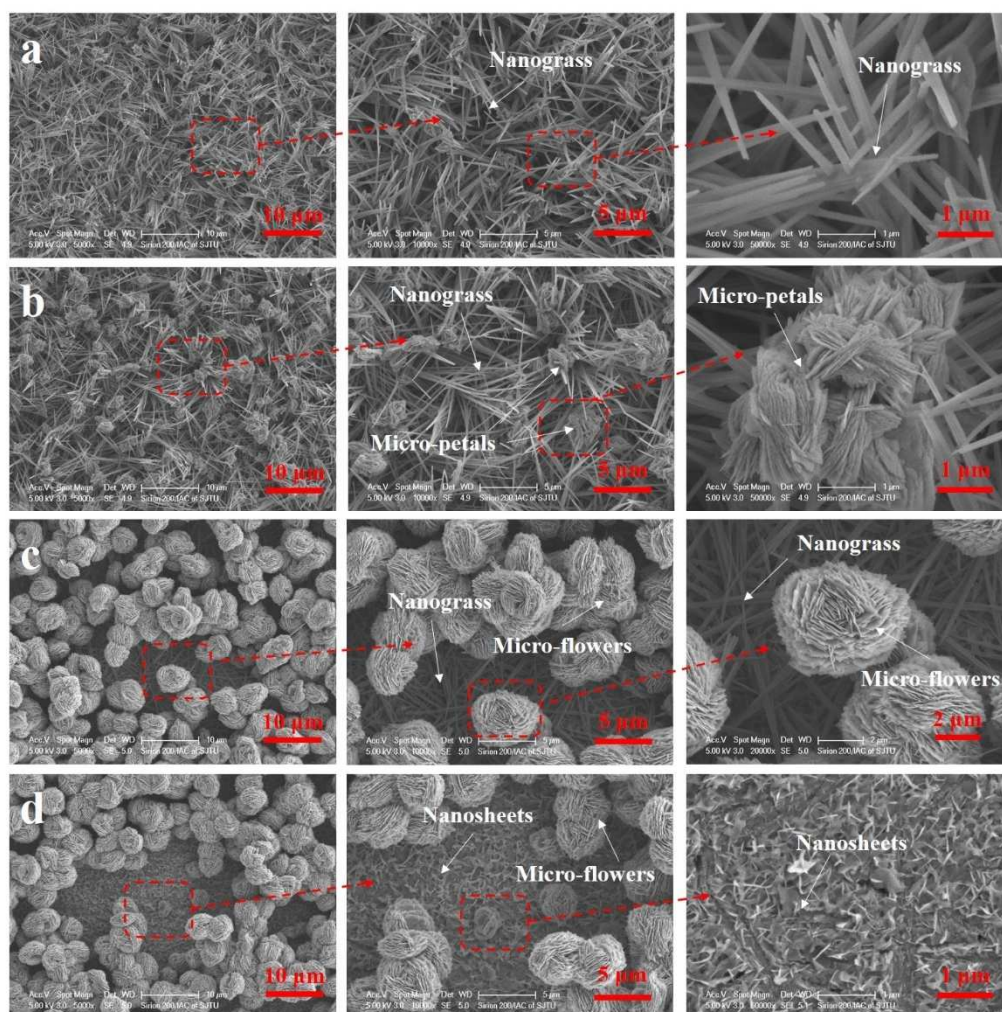
303 The maximum uncertainties of the heat flux, heat transfer coefficient, and wall
 304 superheat for this pool boiling experimental system were $\pm 48 \text{ kW/m}^2$ (8 %), ± 15
 305 $\text{ kW/m}^2 \cdot \text{K}$ (12 %) and $\pm 0.6 \text{ K}$ (9 %), respectively.

306 In addition to the above-mentioned uncertainty analysis, the uncertainty in measuring
 307 the bubble size is also required to be considered. In general, the uncertainty in
 308 measuring the bubble size generates from two factors: dimension conversion when
 309 converting the pixel size into actual size, and measurement error when measuring the
 310 bubble size in the bubble visualization image manually. Due to these two factors, the
 311 total uncertainty in measuring the bubble size is about 10% [59].

312 **3. Results and discussions**

313 **3.1 Surface morphology characterization**

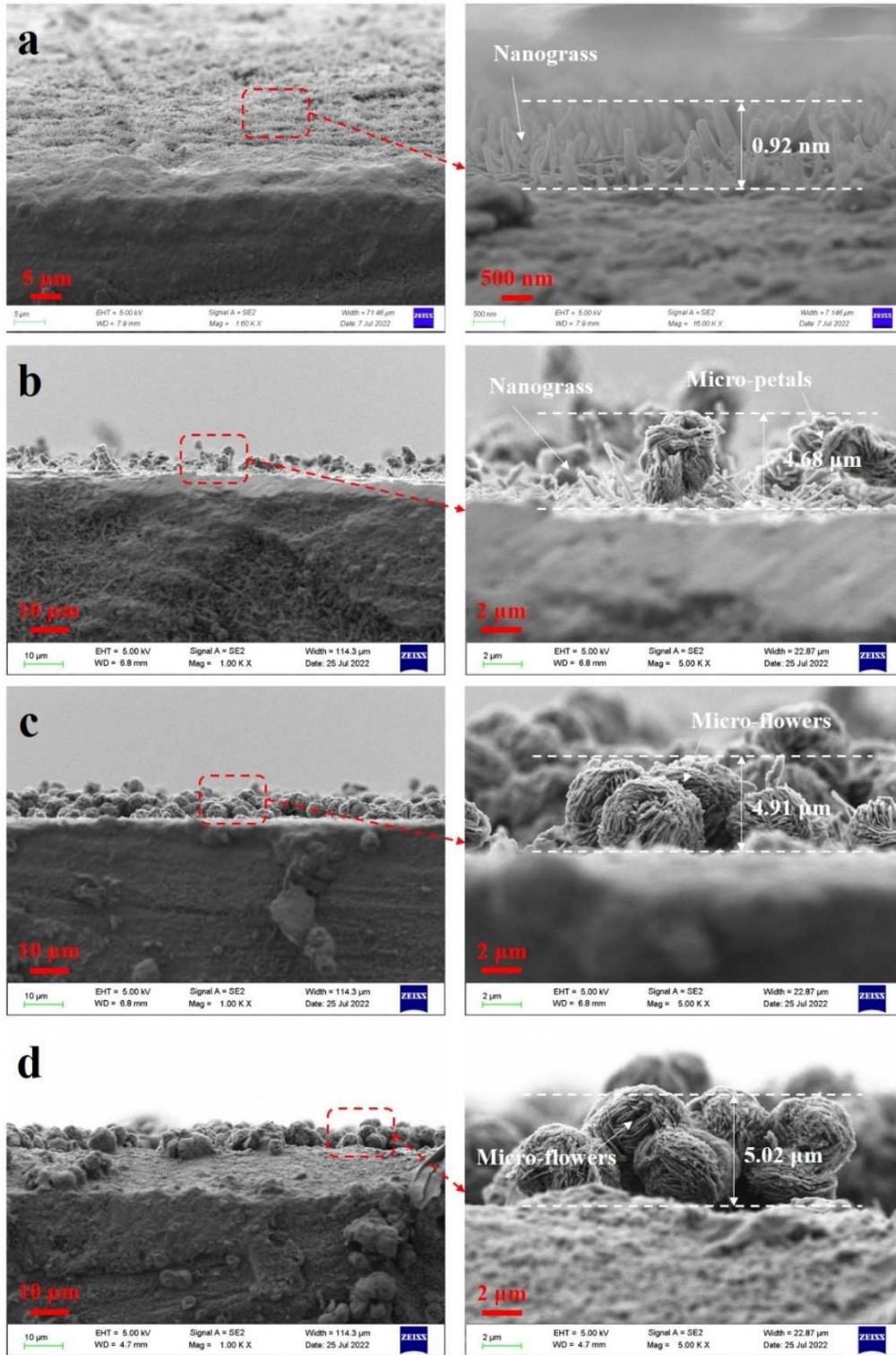
314 **Figure 4** illustrates the top view micro/nanostructures of the different samples. The
 315 immersion time has a clear effect on the growth of the different micro/nanostructures
 316 on the Cu substrate. As shown in **Figure 4a**, when the immersion time was short (3
 317 min), the Cu substrate was mainly covered by uniform and dense nanograss (NG).
 318 Increasing the immersion time to 10 min, nanograss tips can be inter-wined with each
 319 other, forming the micro-petals on the top of the nanograss (**Figure 4b**). This nanograss
 320 and micro-petals (NGMP) coexist almost half and half on the surface. When further
 321 increasing the immersion to 60 min, the micro-petals on the nanograss could be
 322 synthesized to the micro-flowers (NGMF), the Cu substrate was covered by one-layer
 323 nanograss and one-layer micro-flowers, as illustrated in **Figure 4c**. Compared with the
 324 micro-petals, more uniform micro-flowers with a diameter of around $4 \mu\text{m}$ could be
 325 formed on the top of nanograss. The nanograss under the micro flowers were stucked
 326 firmly on the Cu substrate, as seen in **Figure 3c**. As shown in **Figure 4d**, increasing the
 327 immersion time to 120 min, the micro-flowers maintained almost unchanged, and the
 328 Cu substrate was covered by one-layer nanosheets and one-layer micro-flowers
 329 (NSMF).



330

331 **Figure 4.** Top view SEM images of (a) NG, (b) NGMP, (c) NGMF, (d) NSMF.

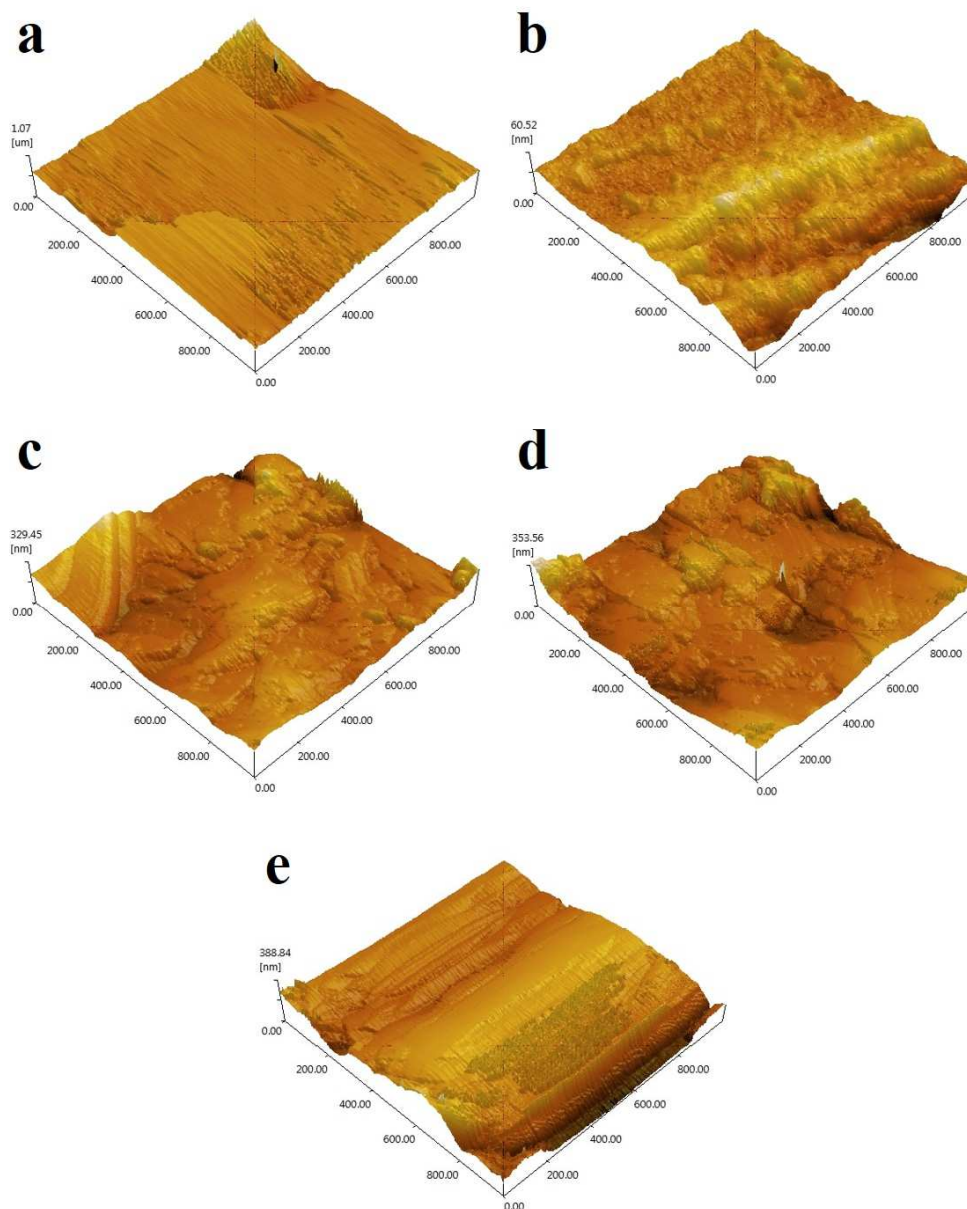
332 **Figure 5** shows the side cross-sectional view SEM images of the different samples.
 333 As shown in **Figure 5a**, the dense nanograss were found to have grown grass-like
 334 structure on the Cu substrate and the thickness of this nanograss is about 0.9 μm . As
 335 seen in **Figure 5b**, micro-petals were found to have grown on the nanograss tips due to
 336 the longer immersion time for synthesis and the thickness of this hierarchical structure
 337 was around 4.7 μm . In **Figure 5c**, the micro-petals synthesized to the larger micro-
 338 flowers and the nanograss seemed to disappear under the micro-flowers because of the
 339 firm stick on the Cu substrate. The thickness of this hierarchical structure was 4.9 μm .
 340 Similar micro-flowers were found in **Figure 5d** to have grown on the Cu substrate and
 341 a large amount of micro-flowers stacked on top of each other, resulting in the larger
 342 thickness of 5.0 μm . Surface roughness of untreated, NG, NGMP, NGMF and NSMF
 343 samples was measured by the AFM method (**Figure 6**). Due to the previous polishing,
 344 the untreated sample exhibited a flat surface (**Figure 6a**). It was clear that the
 345 immersion process caused surfaces to gradually generate and increase rough nanoscale
 346 features (**Figure 6b, c, d and e**).



347

348

Figure 5. Side view SEM images of (a) NG, (b) NGMP, (c) NGMF, (d) NSMF.



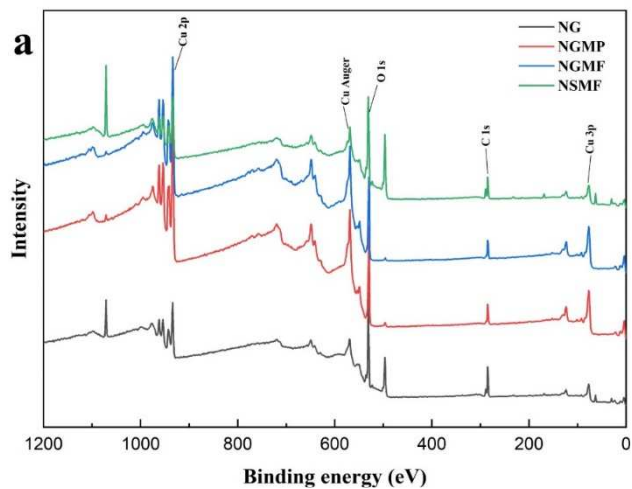
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350 **Figure 6.** AFM images showing the topography of (a) Untreated, (b) NG, (c)
 351 NGMP, (d) NGMF, (e) NSMF.

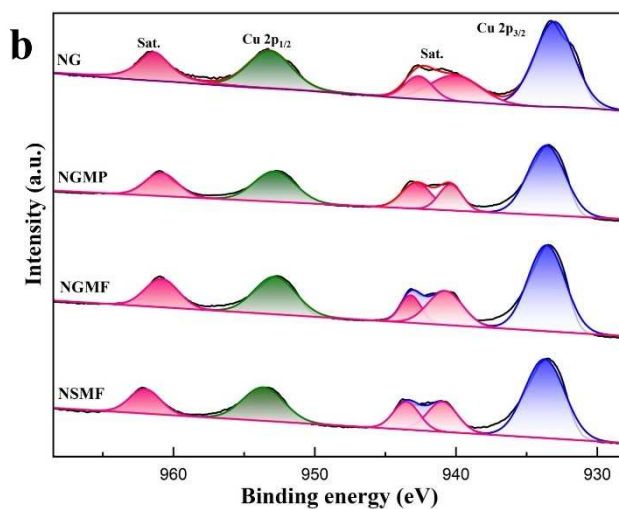
352 **3.2 Surface compositions**

353 **Figure 7a** shows the XPS spectrum of the samples, which indicates that the presence
 354 of the Cu and O as well as the absence of any impurities at various surfaces. As shown
 355 in **Figure 7b**, the high-resolution XPS spectrum of Cu 2p, which includes Cu 2p_{1/2} and
 356 Cu 2p_{3/2} peaks, as well as the Cu²⁺ satellite features, indicating the presence of Cu²⁺ in
 357 the samples. Further analyzing the surface compositions using XRD, as shown in
 358 **Figure 7c**, the peaks shown by as asterisk can be referenced to the cubic phase of copper
 359 (JCPDS card 01-1242). The diffraction peaks for the NG structures were indexed to the
 360 phase of orthorhombic Cu(OH)₂ (JCPDS 14-0420). The XRD diffraction peaks were

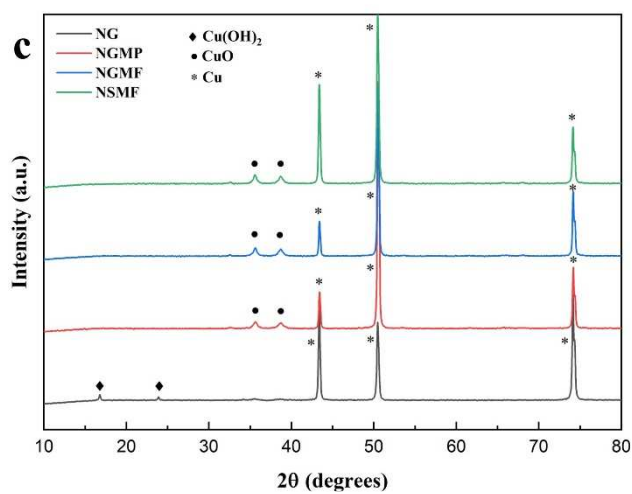
361 for the NGMP, NGMF and NSMF hierarchical structures indexed to the monoclinic
 362 CuO (JCPDS card 5-0611).



363



364



365

366 **Figure 7.** (a) Survey XPS spectrum, (b) High-resolution XPS spectrum of Cu, (c)
 367 XRD patterns of the samples.

368 3.3 Validity of experimental system

369 The boiling curve for a smooth surface is checked with correlations and experiments
 370 from the literature to ensure the validity of the experimental setup. For determining the
 371 dependability of the experimental system, three traditional correlations from the
 372 literature are used to estimate the pool boiling heat transfer on the smooth surface. The
 373 Rohsenow's correlation is expressed as below:

$$374 \frac{c_{p,f} \Delta T_{sat}}{h_{fg}} = C_{sf} \left[\frac{q''}{\mu_f h_{fg}} \sqrt{\frac{\sigma}{g(\rho_f - \rho_g)}} \right]^{-0.33} \left(\frac{c_{p,f} \mu_f}{k_f} \right)^n \quad (10)$$

375 where $C_{s,f} = 0.0152$ and $n = 1$, and Pioro's correlation can be expressed as,

$$376 \frac{q''}{\Delta T_{sat} k_f} \sqrt{\frac{\sigma}{g(\rho_f - \rho_g)}} = C_{sf+} \left\{ \frac{q''}{h_{fg} \rho_g^{1/2} [\sigma g(\rho_f - \rho_g)]^{1/4}} \right\}^{2/3} \left(\frac{c_{p,f} \mu_f}{k_f} \right)^m \quad (11)$$

377 where $C_{sf+} = 1228$ and $m = -1.1$. And CHF value can be calculated by the Zuber's
 378 correlation, which can be expressed as follows,

$$379 q''_{CHF} = \frac{\pi}{24} \rho_g h_{fg} \left[\frac{\sigma g(\rho_f - \rho_g)}{\rho_g^2} \right]^{1/4} \quad (12)$$

380 In addition, Kandlikar proposed a theoretical approach for CHF estimation considering
 381 the surface wettability, which can be expressed as follows:

$$382 q''_{CHF} = h_{fg} \rho_g^{0.5} \left[\sigma g(\rho_l - \rho_g) \right]^{1/4} \frac{1 + \cos \theta}{16} \left[\frac{2}{\pi} + \frac{\pi}{4(1 + \cos \theta)} \right]^{0.5} \quad (13)$$

383 **Figure 8** shows the boiling curve of this study has a closely agreement with the
 384 correlations at the low heat flux. Nevertheless, the CHF value of smooth surface in this
 385 study has the large variation compared with the correlations. In general, the
 386 mechanisms causing CHF can be broadly classified as far-field hydrodynamic theory
 387 and near-field surface property theory. Therefore, there are at least four different models
 388 of CHF: (1) Zuber column Helmholtz-Taylor instability model [60], (2) Lienhard and
 389 Dhir column Helmholtz-Taylor instability model [61], (3) Macrolayer dryout model [56,
 390 62, 63], (4) Hot/Dry spot model [64]. The CHF of smooth surface obtained from the
 391 Zuber's correlation ($CHF_{Zuber's}$) is 100 W/cm^2 . Here note that the Zuber's correlation is
 392 established on the condition of the infinite surface without considering the surface
 393 properties [65]. Considering the size effect on the CHF, Lienhard and Dhir proposed
 394 that the heater size can affect the CHF via reducing the number vapor columns present
 395 on the heater surface. The correlation can be expressed as follows:

$$q_{CHF}'' = 1.14 \times q_{CHF, Zuber's} \times \left(\frac{N_j \times \lambda_D^2}{A_s} \right) \quad (14)$$

where $q_{CHF, Zuber's}$ can be obtained from the Zuber's correlation. N_j is the number of vapor columns on the surface, λ_D is the Taylor instability wavelength and A_s is the heater area. However, this correlation can only be applied when the heater size is much larger than the critical wavelength $\lambda_D = 25$ mm. As listed in Table 2, most actual experimental heater size is smaller than the wavelength λ_D , resulting in one vapor column presented on these surfaces. Therefore, the Zuber's and size effect correlation proposed by the Lienhard and Dhir cannot represent the CHF of the actual experimental surfaces. In addition, the CHF of the surface is affected by the surface wettability, surface roughness, surface morphology, thermal conductivity and et. al [66, 67]. As shown in **Figure 8**, the contact angle of Cu smooth surface in this study is about 109° , exhibiting the hydrophobicity and no liquid wicking and spreading effect on the surface.

Considering the effect surface wettability on the CHF, the CHF of Kandlikar's correlation is 47.7 W/cm^2 because the contact angle of smooth surface in this study is 109° (**Figure 8**). The CHF obtained from the Kandlikar's correction is slightly higher than 42.1 W/cm^2 that we measured. Comparing with the CHF of actual experiments listed in **Table 2**, the surface properties such as the surface wettability and surface roughness are the potential factors of these experimental errors [63]. As a result, the information collected by this experimental system can be trusted.

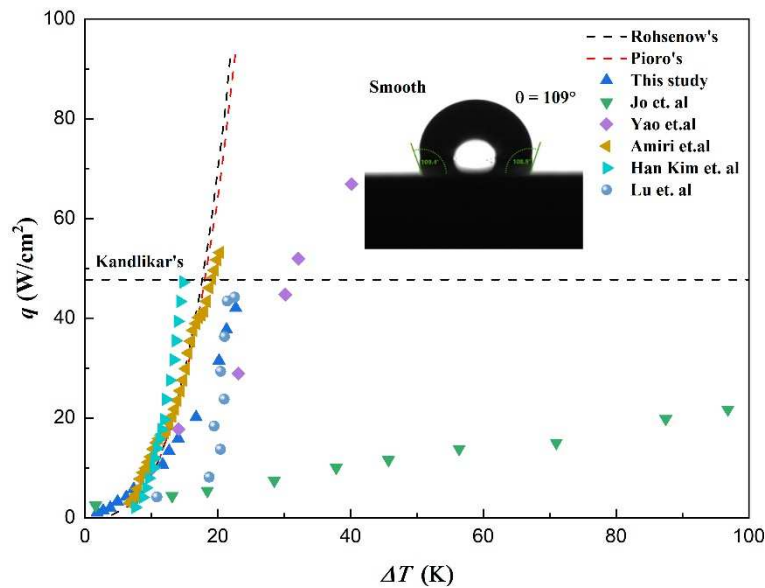


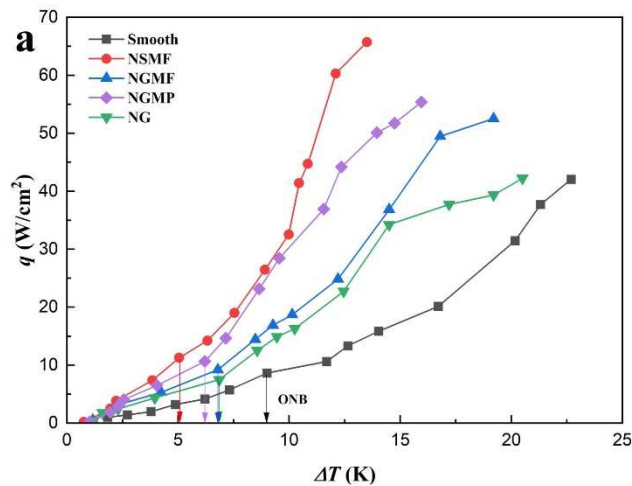
Figure 8. (a) Boiling curve comparisons of water on the smooth surface with Rohsenow's, Piore's correlations, and literatures (b) CHF enhancement for surfaces with various sizes [17, 68-71]

Table 2. CHF values from the literature and present study

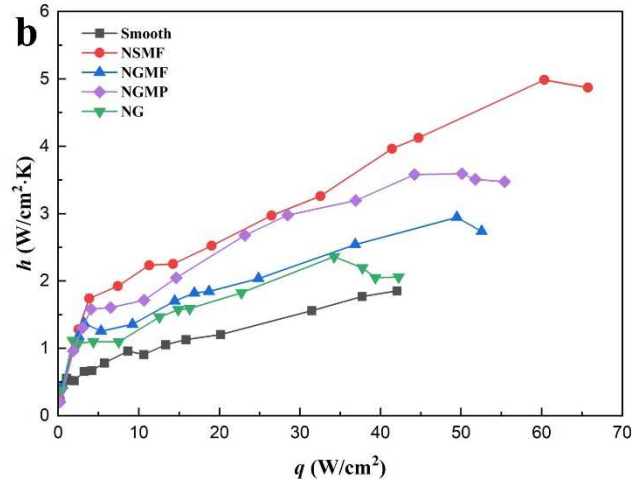
Ref	Material	Shape	D [mm]	CHF [W/cm^2]
[17]	Copper	Circle	30	20
[68]	Silicon	Square	10	65
[69]	Copper	Circle	15	53
[70]	Copper	Square	10	47
[71]	Silicon	Square	20	44.2
This study	Copper	Square	20	42.1

3.4 Pool boiling performance

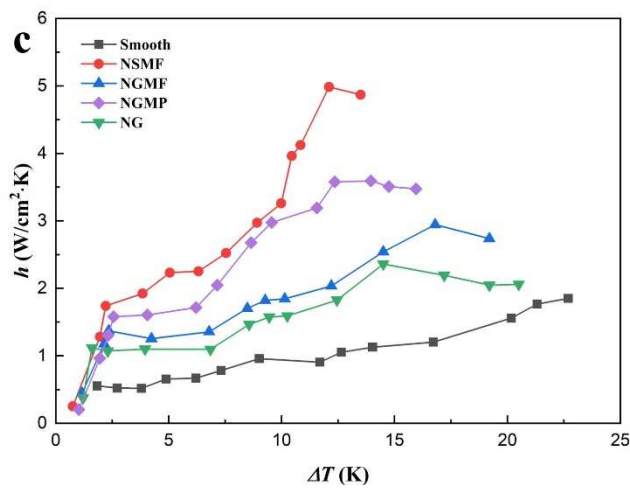
Figure 9a depicts the saturated pool boiling curves on the smooth, NG, NGMP, NGMF and NSMF surfaces at 1 atm. Before the wall reaches a specific value of superheat, it is observed that the heat flux gradually increases, indicating the natural convection process. Beyond the critical point of the wall superheat, the heat flux increases significantly due to the occurrence of the nucleate boiling. The lower ONB, higher CHF, and higher HTC demonstrate that the structured surfaces exhibit significantly improved boiling heat transfer compared the smooth one in Figure 9a, b and c. The ONB among the entire structured surfaces appeared to be decreased from 9 K to less than 7 K, as indicated in Figure 9a (downward arrows). This is because that the micro-/nano-crystal surfaces can improve the nucleate site density, facilitating the bubble generation on the micro/nanostructures. All four structured surfaces exhibited higher CHF when compared with the untreated smooth surface, indicating that the presence of structures allows an improvement of the boiling heat transfer on surfaces. Even though the NG surface had the lowest CHF, it still had a greater CHF than a smooth surface had. Furthermore, the CHF on the smooth surface was lower than that on the hierarchical structures, which included NGMP, NGMF and NSMF, with the values of 55.4, 52.6 and 65.7 W/cm^2 , respectively.



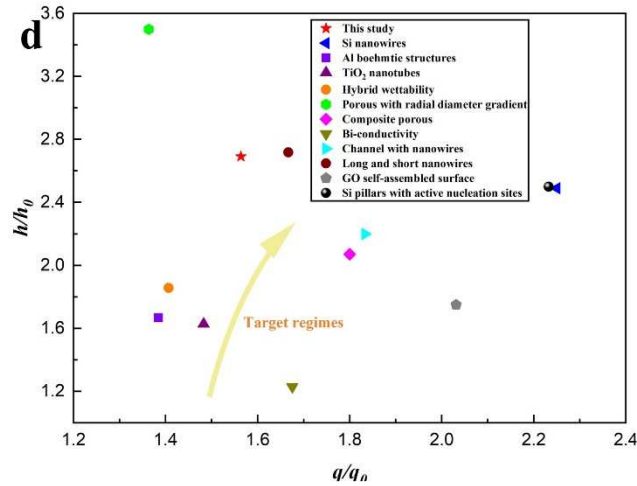
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442 **Figure 9.** (a) Boiling curve of the samples; Heat transfer coefficient for samples as
 443 functions of (b) wall heat flux and (c) wall superheat; Comparison of this study with
 444 other studies of pool boiling on various surfaces.

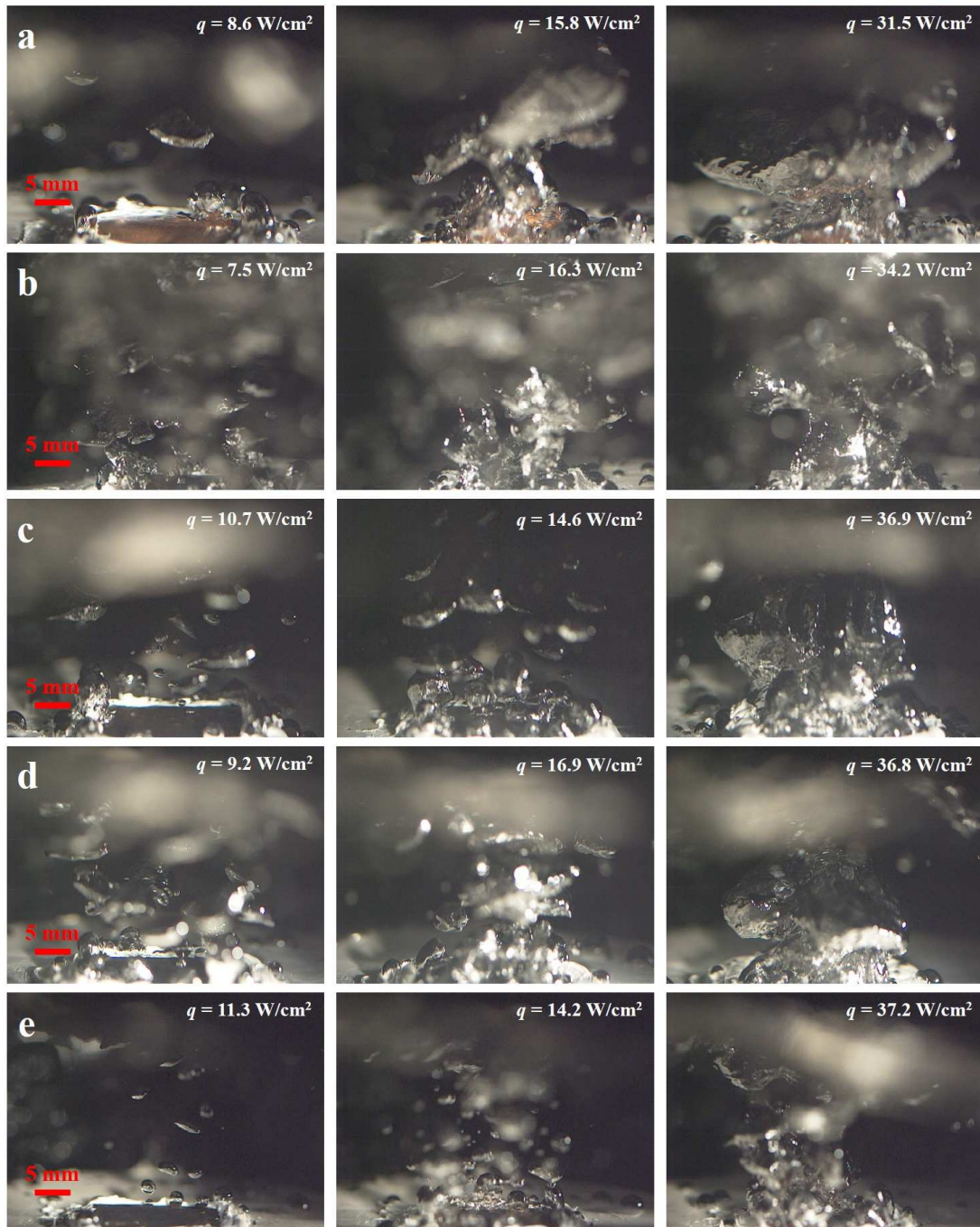
445 According to **Figure 9b**, it was found that the HTC decreased gradually on NG,
 446 NGMF or NGMP surfaces before the onset of the nucleate boiling when the heat flux
 447 is low, in which scenarios the superheat was augmented rapidly. Moreover, the HTC

448 of hierarchical structures involving NGMP, NGMF and NSMF were superior than that
449 of the NG because of the multilevel morphology of the surfaces. Among the three
450 hierarchical structures, the NSMF exhibited the highest HTC of $4.9 \text{ W/cm}^2\text{K}$ compared
451 to the NGMP and NGMF. As shown in **Figure 9c**, due to an early ONB with a much
452 higher HTC and lower wall superheat, the hierarchical surfaces were attractive for
453 thermal management of electronics, batteries and data center to diminish the safety
454 issues raised due to the temperature rise in operational conditions.

455 **Figure 9d** compares the pool boiling performance of various enhancement surfaces
456 investigated in previous literatures under 1 atm [10, 19, 45, 46, 49, 68, 72-76]. For
457 better comparing the boiling performance on the various structured surfaces, we define
458 two parameters h/h_0 and q/q_0 to evaluate the boiling heat transfer. Here h , h_0 , q and q_0
459 represents the HTC of enhanced surfaces, HTC of smooth surface, CHF of enhanced
460 surfaces and CHF of the smooth surface, respectively. As depicted, the most q/q_0 values
461 reported in the literatures were in the range of $1.3 \sim 1.8$, and the q/q_0 of the NSMF was
462 comparable to 1.6. Nevertheless, the NSMF also had the higher HTC at CHF than the
463 other experimental studies. Although few structures like porous with radial diameter
464 gradient had the relatively high h/h_0 , low q/q_0 also could be obtained compared with
465 the data from this study. As comparisons, the hierarchical structures such as long and
466 short nanowires, channel with nanowires, and Si pillars with active nucleation sites
467 provide the similar parameters h/h_0 and q/q_0 compared with the NSMF, which has been
468 proved to be the excellent structures for improving pool boiling heat transfer. As a result,
469 the NSMF can concurrently improve the CHF and HTC, as well as reduce the ONB,
470 which suggests that there is potentials for improving pool boiling heat transfer.

471 **3.5 Bubble visualization and analysis**

472 **Figure 10** shows the bubble dynamics on the smooth, NG, NGMP, NGMF and
473 NSMF surfaces. At a lower heat flux, smaller bubbles appeared on the NG, NGMP,
474 NGMF and NSMF surfaces, and the bubble leaves the surface in nearly spherical shape.
475 While isolated bubble may still be seen on the NGMP and NSMF surfaces, the bubbles
476 above the smooth surface have merged into a single, huge bubble with a mushroom-
477 like form as the heat flux increases. Nevertheless, the NG and NGMF surfaces also
478 have began to bubble coalescence, but there is no mushroom bubble to be observed. It
479 is possible to further activate the number of nucleation sites on the NG, NGMP, NGMF
480 and NSMF surfaces. The bubble coalescence tendency becomes clear as heat flux
481 increases. At the heat flux of about 36 W/cm^2 , the bubbles above the smooth, NG,
482 NGMP and NGMF surfaces have been merged into the large bubble in the mushroom
483 shape, while the merged bubbles are still isolated from the NSMF surface. Therefore,
484 the bubbles can grow and departure from the NSMF surface with faster speed, resulting
485 in the more heat can be carried out from the surface.



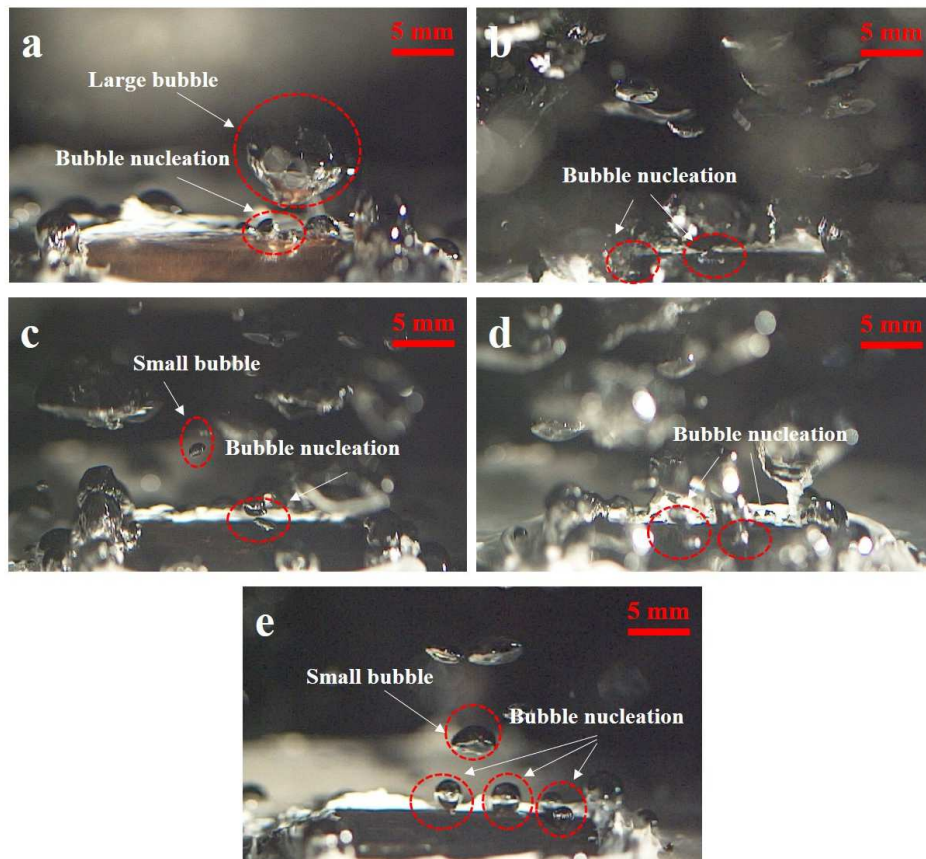
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487 **Figure 10.** Bubble visualization images at $q \approx 10, 15$ and 35 W/cm^2 (a) smooth (b)
 488 NG (c) NGMP (d) NGMF and (e) NSMF

489 **Figure 11** shows the bubble nucleation on the smooth, NG, NGMP, NGMF and
 490 NSMF surfaces at the ONB. As aforementioned, the wall superheat of smooth, NG,
 491 NGMP, NGMF and NSMG surfaces is 9, 6.9, 6.2, 6.8 and 5.1 K at the ONB,
 492 respectively. Comparing with the smooth surface, the structured surfaces including NG,
 493 NGMP, NGMF and NSMF have more bubble nucleation sites or small departure
 494 bubbles, indicating that the NG, NGMP, NGMF and NSMF surfaces can facilitate the
 495 bubble nucleation or bubble departure. Here note that the NSMF surface not only can

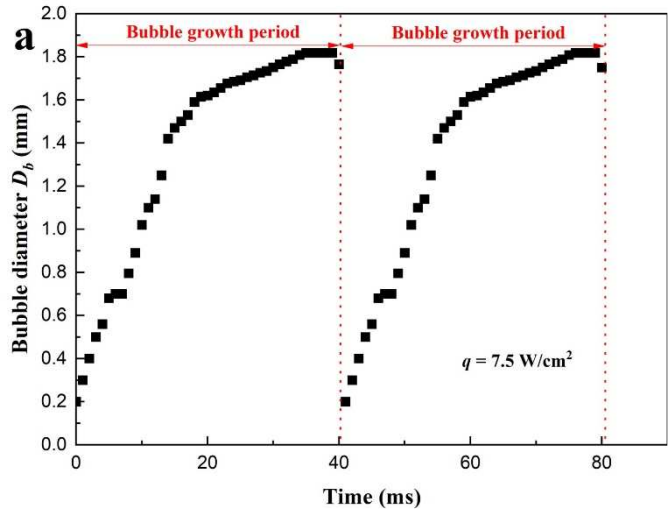
496 provide more bubble nucleation sites but also can facilitate bubble departure comparing
497 with the other surfaces.

498 For further comparing the bubble nucleation on the NG, NGMP, NGMF and NSMF
499 surfaces, we counted the bubble growth and departure periods using high-speed images
500 at the ONB stage. The bubble diameter D_b is calculated by measuring the pixels of high-
501 speed images. Following bubble nucleation, D_b grows significantly during the bubble
502 growth process. Up until the bubbles have a diameter of D_d , the rate of rise in bubble
503 diameter slows. At the same wall superheat, higher heat flux can be obtained on the
504 NSMF surface, and the bubble growth period is shorter than it is on the other surfaces,
505 as shown in **Figure 12a, b, c and d**. This is because that the rapid bubble nucleation
506 and removal from the surface is advantageous for the support of liquid capillary wicking.
507 Nevertheless, the NSMF surface has the smallest departure diameter D_d , while the NG
508 surface has the largest D_d . The bubble departure frequency f increases because it is
509 inversely related to the bubble departure diameter [77]. As a result, the NSMF surface
510 can increase nucleation site density, decrease the bubble diameter and improve the
511 bubble departure frequency, potentially leading to advances in the HTC and CHF.

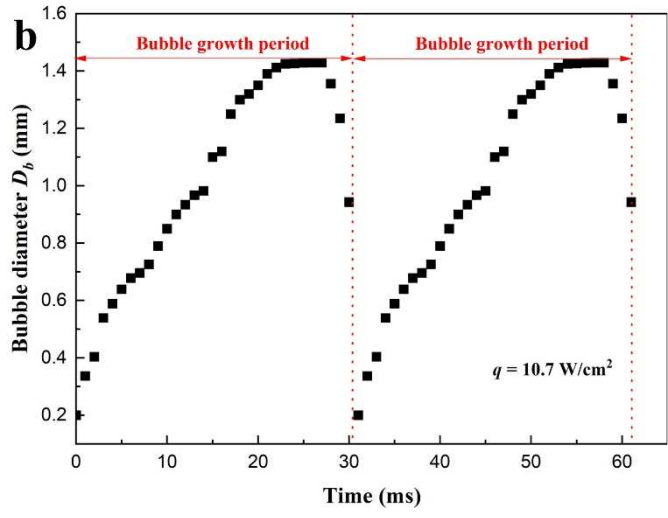


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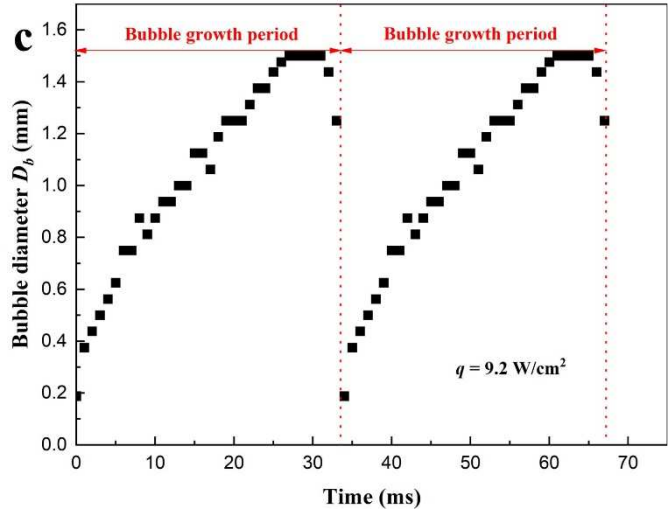
513 **Figure 11.** Bubble nucleation at the ONB for the (a) smooth, (b) NG, (c) NGMP,
514 (d) NGMF, (e) NSMF



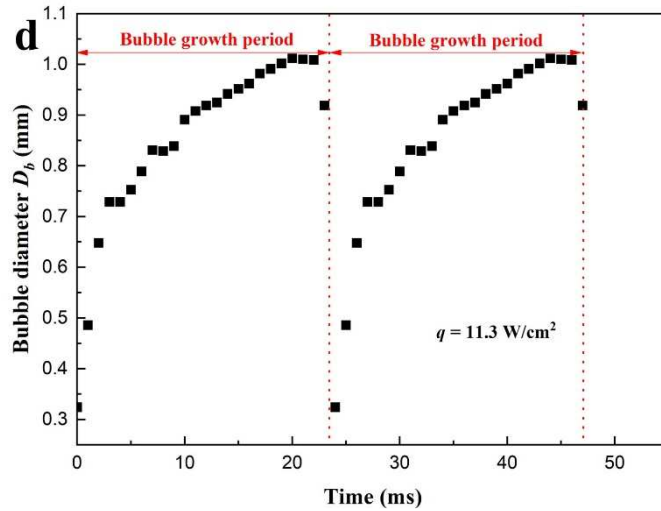
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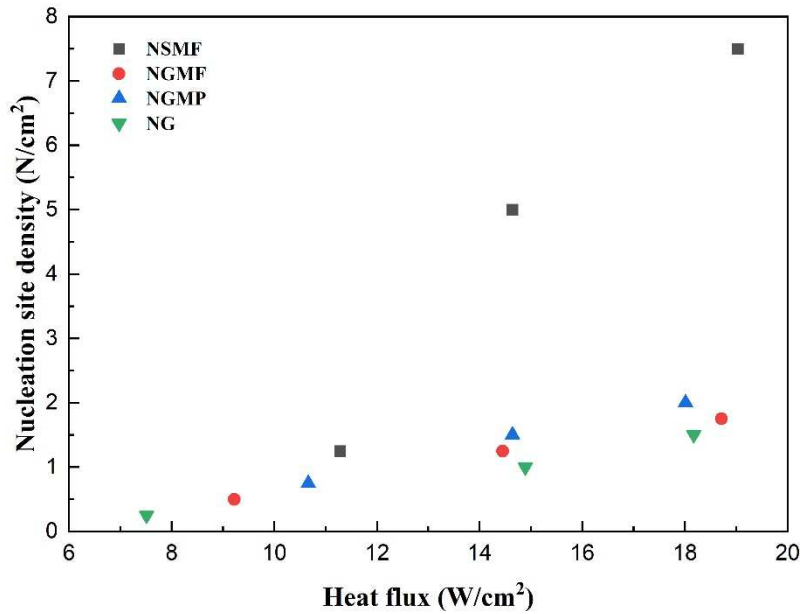
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519 **Figure 12.** Bubble diameters during growth period on the (a) NG (b) NGMP (c)
 520 NGMF (d) NSMF.

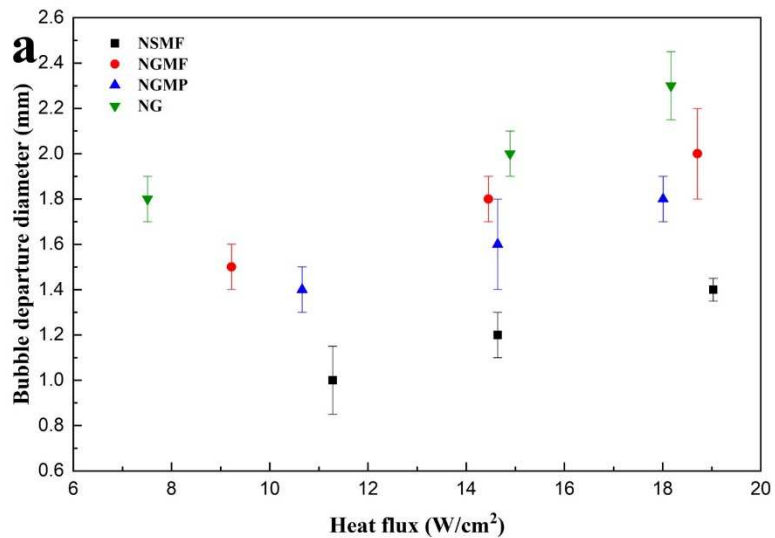
521 As shown in **Figure 13**, the bubble nucleation sites can continue to be activated as
 522 the heat flux increases. We examined the isolated bubbles produced from the surfaces
 523 at low heat flux ($q < 20 \text{ W/cm}^2$) to prevent fluctuation and evaporation of water during
 524 the observation. The bubble nucleation site density is observed to increase as the heat
 525 flux increases on the four different types of surfaces. However, the active nucleation
 526 sites of the NGMF and NGMP only minimally increase, indicating that the hierarchical
 527 structures NGMP and NGMF contribute less to the formation of the active nucleation
 528 sites. For the NSMF, it can be observed that the bubble nucleation site density increases
 529 significantly. As reported, for water at atmosphere, the essential bubble radius ranges
 530 from $0.8\text{-}16 \mu\text{m}$ at the wall superheat of 2 to 40 K [49]. As a result, the microcavities
 531 ($5 \sim 15 \mu\text{m}$) of the NSMF can be exploited as the effective nucleation sites. Because
 532 bubble nucleation is more likely to be activated and the nucleation site density is greatly
 533 increased at low superheat, the ONB can be reduced and the HTC can be enhanced.
 534 Small nucleated bubbles in the microcavities can absorb heat from the surrounding wall
 535 and expand more quickly due to the more activated nucleation sites resulting from the
 536 increased heat flux.



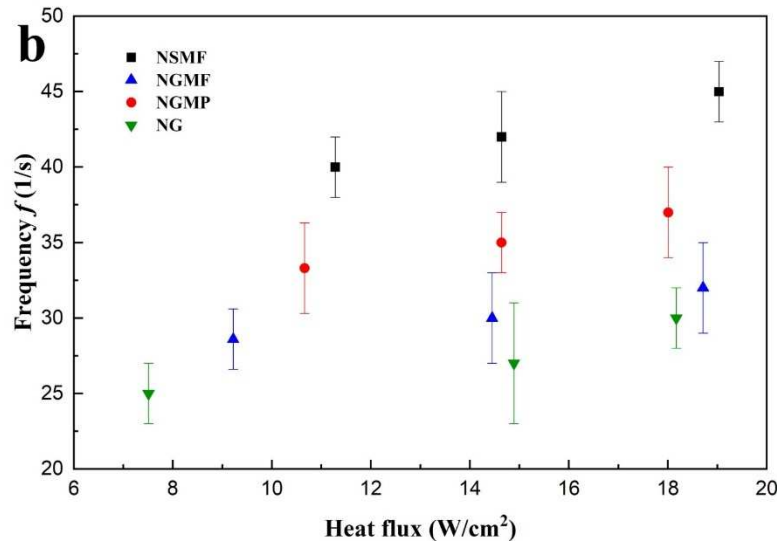
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538 **Figure 13.** Nucleation site density of bubbles with the heat flux on the NG, NGMP,
 539 NSMF and NSMF

540 **Figure 14** shows the average bubble departure diameter and frequency for the NG,
 541 NGMP, NGMF and NSMF surfaces at various low heat flux. As shown in **Figure 14a**,
 542 the bubble departure diameter increases with the increase of heat flux resulting from
 543 the enhanced evaporative rate in the superheated layer. The structure features of the
 544 NSMF result in a substantially smaller bubble departure diameter than those of the NG,
 545 NGMP and NGMF. As shown in **Figure 14b**, the bubble departure frequency on the
 546 NG, NGMP, NGMF and NSMF surfaces exhibits the upward trend with the increase of
 547 the heat flux. Meanwhile, the NSMF has the largest bubble departure frequency of all
 548 the surfaces. This may be because that the NSMF has the fastest liquid replenishing
 549 velocity for accelerating the bubble detachment and offering the liquid to generate the
 550 bubble continuously, thereby increasing the bubble departure frequency.



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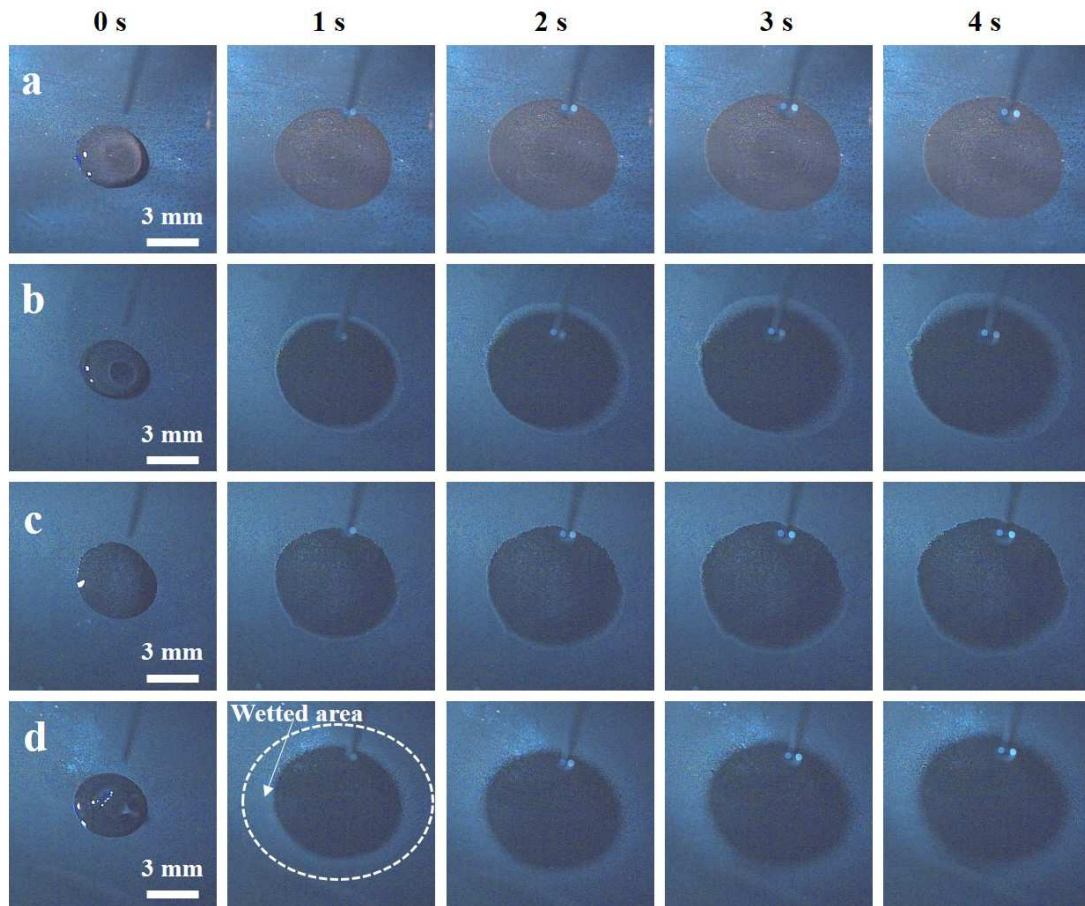


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553 **Figure 14.** (a) Average bubble departure diameter (b) Average bubble departure
554 frequency for the NG, NGMP, NGMF and NSMF

555 **3.6 Effect of capillary wicking on boiling heat transfer**

556 In order to evaluate the wetting properties of the samples, the droplet dynamic
557 behavior on the sample surfaces was recorded. **Figure 15** shows the dynamic behavior
558 of the droplet on the various surfaces. Before dropping the droplet, the initial height of
559 the droplet was 4 mm, and the volume of the droplet was kept at 4 μ L. As the droplet
560 contacted the surface, the contact area between the droplet and the surface increased.
561 As shown in **Figure 15a, b, c** and **d**, the droplet can entirely penetrate the NG, NGMP,
562 NGMF and NSMF surfaces, indicating the super-hydrophilicity of these structured
563 surfaces after the chemical modification. Moreover, after the same period time (4 s) of
564 droplet spreading on the NG, NGMP, NGMF and NSMF surfaces, different wetted
565 areas can be observed various surfaces, indicating different wetting properties of the
566 NG, NGMP, NGMF and NSMF surfaces. As shown in **Figure 15d**, the NSMF surfaces
567 exhibit the largest wetted area after the spreading time of 4 s, which illustrates the most
568 excellent wetting properties among these structured surfaces. Therefore, the effect of
569 the micro/nano structural characteristics on the surface wetting properties can be proved
570 by the droplet dynamic behavior analysis.



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Figure 15. Droplet dynamic behavior analysis (a) NG, (b) NGMP, (c) NGMF, (d) NSMF.

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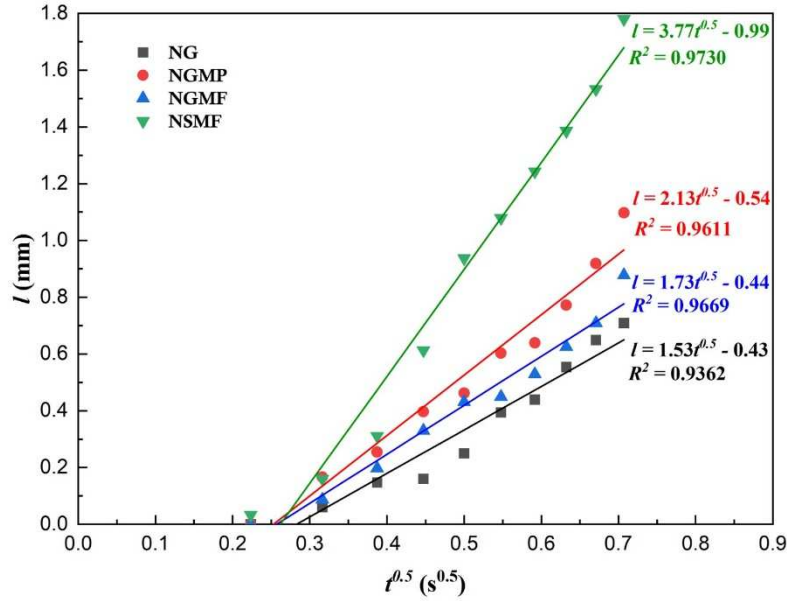
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According to the above-mentioned droplet spreading behavior images, the wicking coefficient which can represent the wicking capability of the surface was calculated according to the equation (1). **Figure 16** shows the wicking coefficient as a function of the square root of time on the NG, NGMP, NGMF and NSMF surfaces in the initial 0.5 s. Larger W value indicates that the droplet spreads faster and the wetted area becomes larger on the structured surfaces. In obvious, a good linear fit between the wicking coefficient and the wicking distance can be observed. The wicking coefficient of the NG, NGMP, NGMF and NSMF is 1.5, 1.7, 2.1 and 3.8 mm/s^{0.5}.



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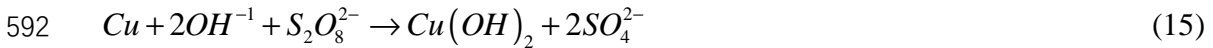
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Figure 16. Wicking distance with square root time

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To further explore the effect of the oxide presence on the surfaces' wicking capability, we conducted the EDS analysis on the NG, NGMP, NGMF and NSMF surfaces. As shown in **Figure 17a, b, c, and d**, the O content of the NG, NGMP, NGMF and NSMF is 32.5%, 24.9%, 24.7%, 24.3%, respectively. Meanwhile, the Cu content of the NG, NGMP, NGMF and NSMF is 35.5%, 58.3%, 58.6%, 61.3%, respectively. With the increase in the reaction time, the O content decreases and the Cu content increases, indicating that more Cu^{2+} will produce with the presence of O. According to the reaction process as follows [78]:

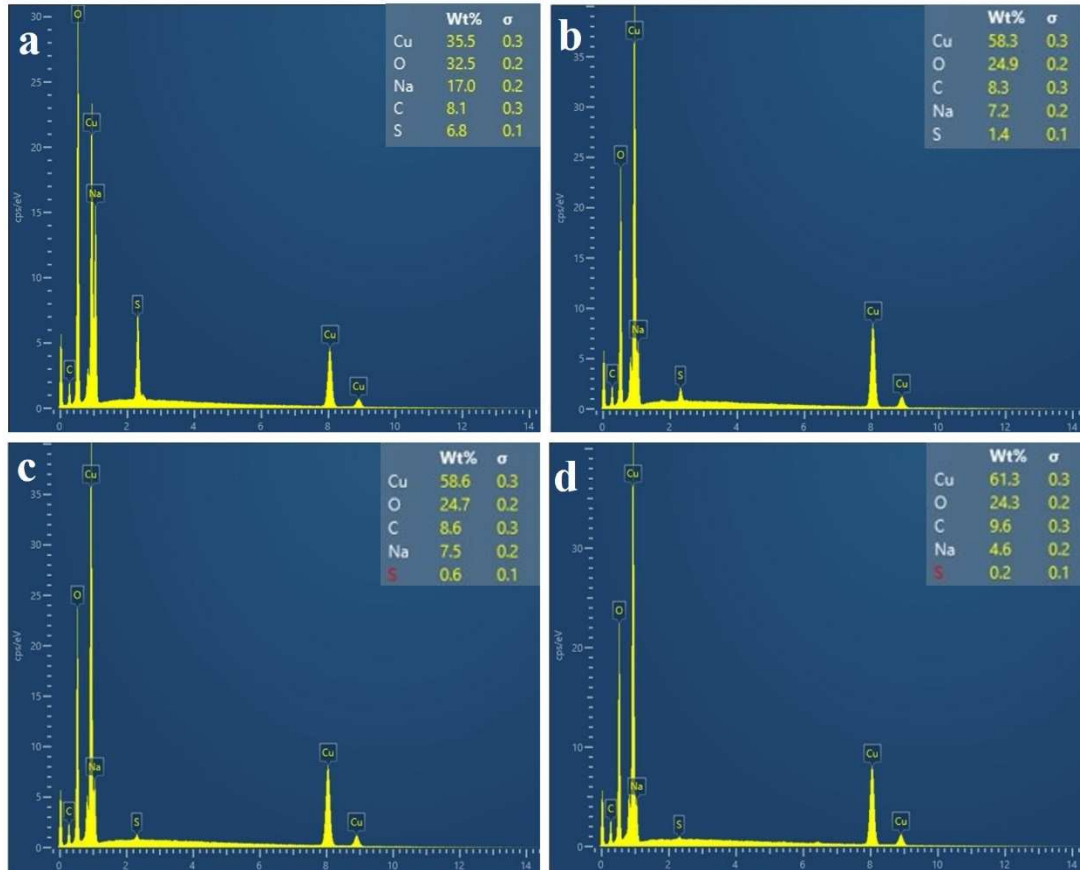
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When the reaction time is shorter, NG surface still has more Na and S contents, indicating that OH^{-} and $\text{S}_2\text{O}_8^{2-}$ are not fully participating in the reaction. Thus, the O and Cu contents are smaller on the NG surface. Further increasing the reaction time, more O content can participate in the reaction, and the $\text{Cu}(\text{OH})_2$ on the NG converts into the CuO. The O and Cu contents can maintain almost unchanged when the NG structure converts into the other hierarchical structures. This is attributed to the fact that more Cu^{2+} will produce with the presence of more O. When converting into the CuO, the NGMP, NGMF and NSMF become more stable, resulting in no significant formation of the Cu^{2+} . Nevertheless, different wicking performance can be observed on the NG, NGMP, NGMF and NSMF surfaces. Therefore, the presence of oxide has almost no effect on the wicking.

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Figure 17. EDS analysis (a) NG, (b) NGMP, (c) NGMF, (d) NSMF

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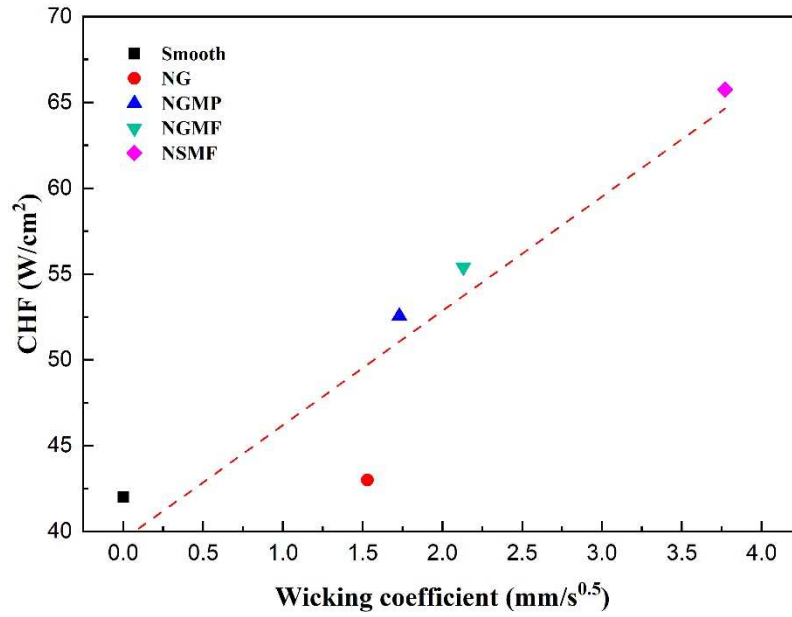
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Figure 18 shows the relationships between the CHF and W . The wicking coefficient W is zero on the smooth surface because of no micro/nanostructures fabricated on its surface. The CHF and wicking coefficient have a linear relationship, which suggests that the W alone determines the CHF on the NG, NGMP, NGMF and NSMF. In obvious, the CHF of the NG surface is a little higher than that of the smooth surface. This may be attributed to two factors. First, the $\text{Cu}(\text{OH})_2$ on the NG surface is easily to decomposed into the stable CuO [79], which may destroy the nanograin structure on the NG surface. Second, the thickness of $\text{Cu}(\text{OH})_2$ nanograin on the NG is very thin ($0.9 \mu\text{m}$), which will be destroyed during the pool boiling process, resulting in exposing smooth copper surface directly and deteriorating the CHF. Comparing with the NG structures, the hierarchical structures including (NGMP, NGMF and NSMF) have the significantly high CHF. This is attributed to the fact that the NGMP, NGMF and NSMF structures have the higher wicking capability and multilevel roughness [55]. As the wicking coefficient increases, the CHF of NGMF, NGMP and NSMF structure increases, which shows that the CHF augmentation might be caused by a wicking surface with improved wicking capabilities. After the growth of micro/nanocrystals on the smooth surface, the microcavities can be fabricated on the surface for bubble nucleation. Additionally, capillary-driven constant liquid replenishment to the dry-out region can improve the CHF. As a result, the NSMF's largest liquid capillary wicking

626 capability produces the highest CHF.



627

628 **Figure 18.** Relationship between CHF and wicking coefficient W

629

629 3.7 Pool boiling enhancement mechanism

630 In the nucleate boiling process, the convective heat transfer can be predicted by the
 631 bubble departure diameter D_d , nucleation site density N_a and bubble departure
 632 frequency f . The equation can be expressed as follows [80]:

$$633 \quad HTC = 2\sqrt{\pi k_l \sigma C_p} \sqrt{f D_d^2 N_a} \quad (17)$$

634 where k_l , C_p , and σ are the liquid thermal conductivity, specific heat and surface tension,
 635 respectively. As aforementioned, the minimum cavity radius required for activation is
 636 determined by

$$637 \quad R_c = \frac{2\sigma T_{sat}}{\rho_l h_{fg} \Delta T} \quad (18)$$

638 NSMF can provide the suitable size range (5 ~ 15 μm) to act as the cavity for bubble
 639 nucleation according to **Figure 4d**. In addition, the cavity radius will become smaller
 640 as the wall superheat increases at higher heat flux. Here note that a wall superheat of 5
 641 K refers to a cavity diameter of 13 μm , while a higher wall superheat of 20 K decreases
 642 the cavity diameter of 3 μm , which is almost the pore size of the NSMF. Therefore, at
 643 the low heat flux, more bubbles can be activated, and the nucleation site density can be
 644 significantly increased, resulting in the reduced ONB and the increase in the HTC. As
 645 shown in **Figure 19**, with the increase in the heat flux, more micro pores as nucleation
 646 sites are activated, small nucleated bubbles in the cavities can absorb heat from the
 647 surrounding wall (micro-flower structures) of the cavities, resulting in the faster bubble

648 growth rate. In addition, the surrounding wall (micro-flower structures) can limit the
649 bubble expansion, accelerating the bubble departure and improving the bubble
650 departure frequency.

651 At the high heat flux, liquid supply performance which is determined by the capillary
652 pressure significantly affect the pool boiling performance. The capillary pressure can
653 be expressed as follows:

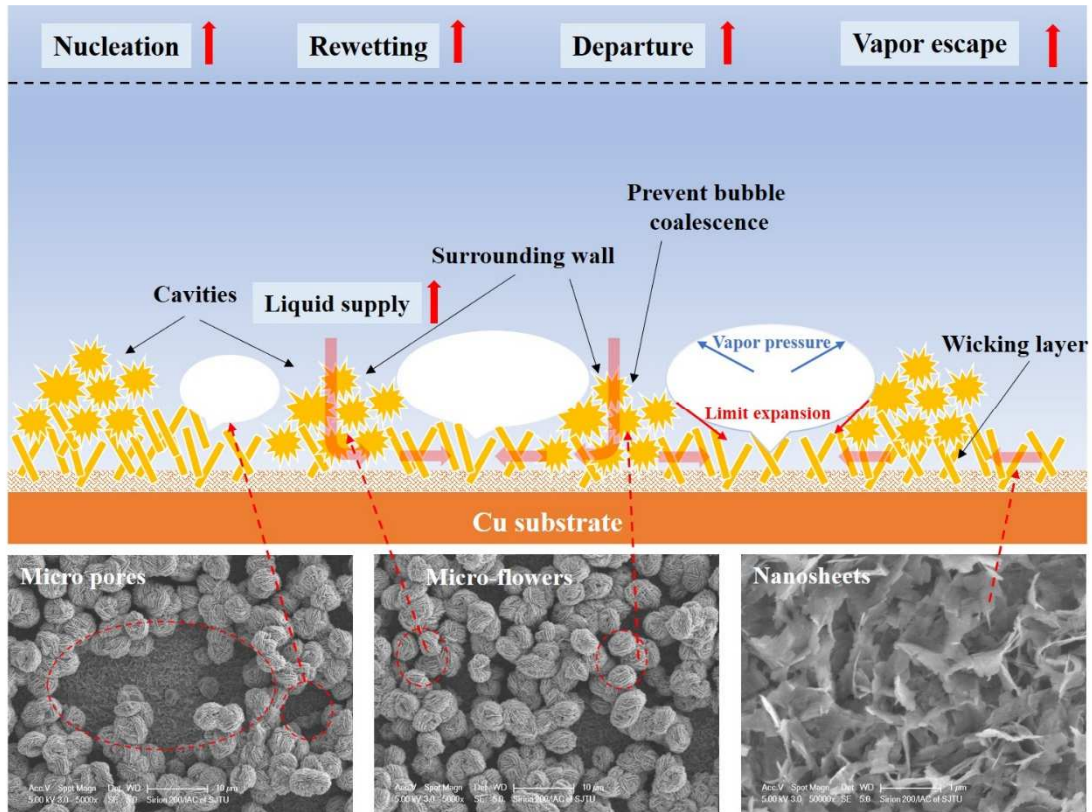
$$654 \quad \Delta P = 4\sigma / D_p \quad (19)$$

655 where σ is the surface tension and D_p is the pore size. It is clear that smaller pores can
656 facilitate the liquid to rewet the nucleation, delaying the formation of hot spots and
657 enhancing the CHF. However, the performance of vapor escape is also the other
658 important factor to influence pool boiling performance. The vapor mass rate can be
659 defined as follows [81]:

$$660 \quad m = \frac{\pi}{12} \left(\frac{\rho_v \sigma}{\mu_v} \right) \left(\frac{\varepsilon D_p^3}{\delta} \right) \quad (20)$$

661 where ρ_v , μ_v , ε and δ is the vapor density, vapor viscosity, porosity and thickness.
662 According to the equation (20), larger pores can facilitate the vapor mass rate, resulting
663 in better vapor escape. Therefore, according to the above-mentioned equations (19) and
664 (20), the optimized pore size can simultaneously enhance the liquid supply and the
665 vapor escape. As shown in **Figure 19**, the NSMF structure is composed of nanosheet
666 bottom layer and micro-flower top layer. Micro pores can also act as the vapor escape
667 channel to facilitate the bubble departure. Meanwhile, the liquid supply can rewet the
668 cavities (micro pores) immediately through the surrounding wall with small pores, and
669 fast liquid supply can also facilitate the bubble departure. Furthermore, the surrounding
670 wall can prevent the bubble coalescence when bubbles generated from the adjacent
671 micro pores become large enough to some extent. On the bottom, nanosheet structure
672 with thin thickness can serve as the wicking layer to drive the liquid to rewet to the
673 nucleation sites for delaying the formation of hot spots, resulting in the CHF
674 enhancement.

675 Hence, at the low heat flux, the NSMF structure can provide more nucleation sites
676 for reduced ONB and facilitate bubble departure for improved HTC, while NSMF
677 structure can also facilitate the vapor escape and improve liquid supply at the high heat
678 flux. The integration of the micro pore, micro-flower and nanosheet structures
679 contributes to the better boiling performance comparing with the other structures
680 including NG, NGMP and NGMF.



681

682

Figure 19. Enhancement mechanism of boiling heat transfer on the NSMF.

683

4. Conclusions

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In this study, the enhancement in pool boiling heat transfer of the hierarchical micro/nanocrystal structures on the Cu substrate in DI water was specifically examined under a wide range of heat fluxes. Based on the experiments, the following conclusions can be drawn:

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(1) The SEM, XRD, AFM and XPS methods were used to characterize the morphology and investigate the chemical content of these hierarchical micro/nanocrystals. With the in-situ growth on the Cu substrate, the nanoneedles (NG), nanoneedles and micro-petals (NGMP), nanoneedles and micro-flowers (NGMF) and nanosheets, nanosheets and micro-flowers (NSMF) can be synthesized. The thickness of the NG, NGMP, NGMF and NSMF are 0.9, 4.7, 4.9 and 5.0 μm , respectively. The compositions of the NG, NGMP, NGMF and NSMF are $\text{Cu}(\text{OH})_2$ and CuO according to the XRD analysis.

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(2) Compared with the capillary wicking on the NG, NGMP, NGMF and NSMF surfaces, the NSMF surface has the largest capillary wicking number W_i of 0.3, which is higher than those values on the NG, NGMP and NGMF surfaces. The experiments indicate that the surface roughness and morphology are responsible for the improvements in capillary wicking.

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(3) Compared with the smooth and hierarchical structures, NSMF surface has the

702 largest CHF of 65.7 W/cm², which implies a significant improvement of 56.5%,
703 55.8%, 18.6% and 25.1% compared to the smooth surface, the NG, NGMP, and
704 NGMF, respectively. Meanwhile, the HTC can be augmented to 4.9 W/cm²K on the
705 NSMF surface, which is larger than that of other surfaces in the same conditions.
706 Moreover, superheat for the ONB on the NSMF is only 5.1°C, which is 43.3%,
707 26.1%, 17.7% and 25.0% lower than those on the smooth surface, the NG, NGMP,
708 NGMF, respectively.

709 (4) These significant improvements are due to the enhanced capillary wicking and
710 increased bubble releasing frequency in the boiling phenomenon caused by the
711 special hierarchical morphology of the NSMF surface. As a result, the dynamics by
712 the liquid circulation is remarkably strengthened while the resistance by the
713 coalescence of bubbles and formation of vapor film is largely reduced. Nonetheless,
714 the surface wettability and the measurement accuracy of the experimental apparatus
715 are also pivotal factors that need to be optimized. Further experimental
716 investigations are to be completed towards a breakthrough of the heat transfer
717 performance on such hierarchical structures.

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